

# On the Ytterbium Valence and the Physical Properties in Selected Intermetallic Phases

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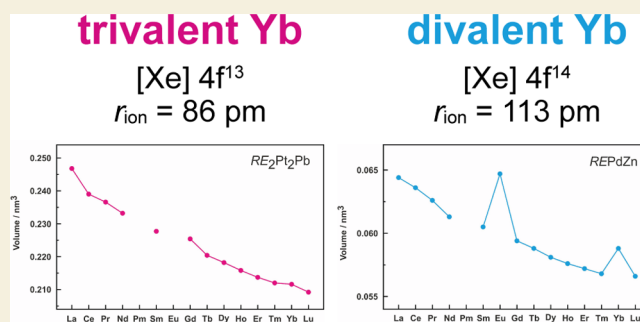
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**ABSTRACT:** The present review summarizes important aspects of the crystal chemistry of ytterbium-based intermetallic compounds along with a selection of their outstanding physical properties. These originate in many cases from the ytterbium valence. Different valence states are possible here, divalent ( $4f^{14}$ ), intermediate-valent, or trivalent ( $4f^{13}$ ) ytterbium, resulting in simple diamagnetic, Pauli or Curie–Weiss paramagnetic, or valence fluctuating behavior. Especially, some of the  $Yb^{3+}$  intermetallics have gained deep interest due to their Kondo or heavy Fermion ground states. We have summarized their property investigations using magnetic and transport measurements, specific heat data, NMR, ESR, and Mössbauer spectroscopy, elastic and inelastic neutron scattering, and XAS data as well as detailed thermoelectric measurements.

**KEYWORDS:** ytterbium, intermediate valence, Kondo effect, heavy Fermion,  $^{170}Yb$  Mössbauer spectroscopy,  $^{171}Yb$  NMR, ESR, XAS, XPS, valence fluctuations



Ytterbium is one of the few rare earth elements that is prone to valence instabilities in intermetallic compounds. In the rare earth series, many examples are known for cerium, where the interplay between the  $4f^1$  configuration for  $Ce^{3+}$  and the empty  $f$  shell for tetravalent cerium led to a huge number of compounds with broadly varying magnetic and electrical properties, such as long-range magnetic ordering (antiferro-, ferro-, or ferrimagnetism), Kondo-type behavior, heavy Fermion behavior as well as static or dynamic intermediate cerium valence. These many facets of structure–property relationships of cerium intermetallics have been summarized in review articles.<sup>1–6</sup> While only a few examples are known for divalent samarium (the trivalent state is usually observed),<sup>7</sup> many europium compounds have been studied with respect to their valence behavior.<sup>8–10</sup> Most of these compounds exhibit the stable half-filled  $f$  shell ( $4f^7$  configuration), and consequently, divalent europium is the usual case for intermetallics along with paramagnetism and long-range magnetic ordering, sometimes at comparatively high ordering temperatures. The scarce case of trivalent europium in intermetallic phases has recently been reviewed.<sup>10</sup>

The present review focuses on intermetallic ytterbium compounds. Here, the configurations  $4f^{14}$  (filled  $f$  shell; divalent ytterbium; diamagnetic) and  $4f^{13}$  (paramagnetic trivalent ytterbium) are the edge states, and static and dynamic ytterbium valence states are observed in-between. Especially, the trivalent ytterbium compounds have attracted high attention in solid-state chemistry and physics, since the  $4f^{13}$

configuration is the  $4f$  hole analogue to  $Ce^{3+}$ . Thus, the ground state properties of these ytterbium-based phases are highly sensitive to the amount of  $4f$  electron (hole)–conduction electron hybridization (Kondo effect).<sup>11–14</sup> This influences the Ruderman–Kittel–Kasuya–Yosida (RKKY) exchange interactions and consequently the magnetic ordering. The broad analogy to cerium intermetallics is especially interesting with respect to intermediate-valence and potential heavy Fermion materials.<sup>15–17</sup>

Materials science can only work with an excellent interplay of solid-state physics and solid-state chemistry, whereby detailed phase analyses and high-quality sample preparation along with precise structure determination are the indispensable prerequisites for reliable property measurements. Herein we review these features for the field of intermetallic ytterbium compounds with a strong emphasis on the ytterbium valence behavior.

The review is written from a solid-state chemist’s point of view. Bearing the enormous number of compounds<sup>18</sup> and property measurements in mind, it is simply not possible to

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include all known intermetallic ytterbium compounds and all references into such an overview (e.g., > 350 references for  $\text{YbRh}_2\text{Si}_2$  in SciFinder<sup>19</sup>). Herein we mainly concentrate on the role of the ytterbium valence in intermetallics and underpin these peculiarities with selected examples. Any missing compound or reference is unintentional.

## SYNTHESES AND CRYSTAL GROWTH

The peculiar electronic structure of ytterbium has drastic effects on its physical properties, when compared with the neighboring rare earth elements. Its divalent nature already in the elemental state leads to drastically reduced melting (1097 K) and boiling (1466 K) points.<sup>20</sup> Arc-melting is a broadly used synthesis technique for intermetallic samples,<sup>21</sup> however, drastic evaporation losses can occur in the cases of ytterbium-based synthesis, when reacting ytterbium with the high-melting noble metals, e.g., platinum (mp 2025 K) or iridium (mp 2683 K). To give an example, during an arc-melting synthesis of  $\text{YbRh}_2\text{Si}_2$ ,<sup>22</sup> the ytterbium would largely evaporate before the rhodium melts. Thus, only a few ytterbium intermetallics were prepared by arc melting. Representative examples are  $\text{YbNiAl}_4$ <sup>23</sup> or  $\text{Yb}_3\text{Rh}_9\text{Si}_2\text{Sn}_3$ .<sup>24</sup> For the precise reaction conditions, we refer to the individual publications. In the present chapter, we summarize the different techniques that find application.

Broader phase analytical studies were conducted by simple, direct reaction of the elements in sealed evacuated fused silica ampules. This is possible for almost all  $\text{Yb}_x\text{T}_y\text{X}_z$  samples with  $X$  = elements of the group 13, 14, or 15. The inner surface of the silica ampules can be graphitized (thermal decomposition of acetone) for passivation, or alumina crucibles can be used as inert container material. The reactions and annealing sequences are usually carried out in tube or muffle furnaces. These reactions are not that far from the classical ceramic method<sup>25</sup> for the preparation of solids. Often the samples are still multiphase after the first annealing cycle. Thus, repeated grinding, pressing of pellets, and reannealing is necessary in order to obtain samples that are at least phase-pure on the level of powder X-ray diffraction. A critical feature is the repeated grinding, since the increasing surface area might lead to surface oxidation and thus  $\text{Yb}_2\text{O}_3$  contamination (vide infra). This ceramic technique was frequently used for the synthesis of ytterbium-based ternary phosphides and arsenides.<sup>26</sup> The best way to avoid these  $\text{Yb}_2\text{O}_3$  contaminations is the use of distilled ytterbium and working under purified inert gas.

For physical property measurements, high sample purity is required. Nowadays, the intermetallic ytterbium phases are prepared in sealed (arc-welded<sup>27</sup> under argon or helium atmosphere) high-melting metal tubes, usually niobium or tantalum, but molybdenum can also find application. This ensures largely inert crucible conditions and prevents evaporation losses of ytterbium. For the reaction, the metal tubes can be sealed in silica tubes (oxidation protection), and the annealing sequence can be run in a tube or muffle furnace. Much better results are obtained through induction melting and subsequent annealing.<sup>28,29</sup> Good quantities of X-ray pure polycrystalline samples are available by this technique.

Direction-dependent property measurements can only be performed on high-quality single crystals. This deserves sophisticated crystal growth conditions. The flux growth technique<sup>30–33</sup> is meanwhile well established in the field of crystal growth of intermetallic phases. Usually the low-melting  $p$  block elements aluminum, gallium, indium, tin, and bismuth

find application as flux medium. Two strategies for crystal growth are possible. The easy way is the self-flux technique (the so-called reactive flux), where the low-melting  $p$  element is a component of the product and flux medium at the same time. Otherwise, one can use an inert flux (the so-called nonreactive flux), which does not enter the product crystals. There are some prerequisites for targeted crystal growth: (i) the flux medium should be inert toward the crucible material and (ii) the excess flux should easily be separable from the product crystals.

Typical examples for flux-grown crystals are  $\text{YbRh}_2\text{Si}_2$  and  $\text{YbIr}_2\text{Si}_2$ ,<sup>34</sup> or  $\text{Yb}_2\text{CuGe}_6$  and  $\text{Yb}_3\text{Cu}_4\text{Ge}_4$ ,<sup>35</sup> which form in liquid indium. Also,  $\text{Yb}_2\text{AuGe}_3$  crystals can be grown from liquid indium with up to 2 mm edge size.<sup>36</sup> The excess flux can be dissolved in diluted hydrochloric acid. Similar flux growth experiments are feasible with aluminum and gallium as the flux medium.<sup>37</sup> A interesting example for a nonreactive flux is the growth of  $\text{YbRh}_6\text{P}_4$  crystals in liquid bismuth.<sup>38</sup> Besides dissolution of the flux after successful crystal growth, the liquid flux can also be separated by centrifugation at elevated temperatures. Here are two possible ways: (i) a second crucible filled with quartz wool or a quartz wool plug can be placed (reversed) on top of the one containing the starting materials<sup>30</sup> or (ii) a frit-disc crucible set (“Canfield crucible set”) can be utilized.<sup>39</sup>

Finally, we turn back to the classical textbook examples for crystal growth. Due to the high vapor pressure of molten ytterbium, the Czochralski technique cannot be used. The Bridgman technique (where again sealed tubes are used) was successfully applied for crystal growth of the copper-rich ytterbium compounds  $\text{YbCu}_{4.25}$  and  $\text{YbCu}_{4.4}$ .<sup>40</sup> Tantalum was used as container materials, and the crucible lowering speed at the high-frequency coil was  $3 \text{ mm h}^{-1}$ . Crystals for diffraction experiments were finally separated from different slices that were prepared from the bulk crystal via spark erosion. Further examples for the Bridgman crystal growth are the germanides  $\text{Yb}_2\text{Ru}_3\text{Ge}_4$ ,<sup>41</sup>  $\text{YbPdGe}$ , and  $\text{YbPtGe}$ .<sup>42</sup>

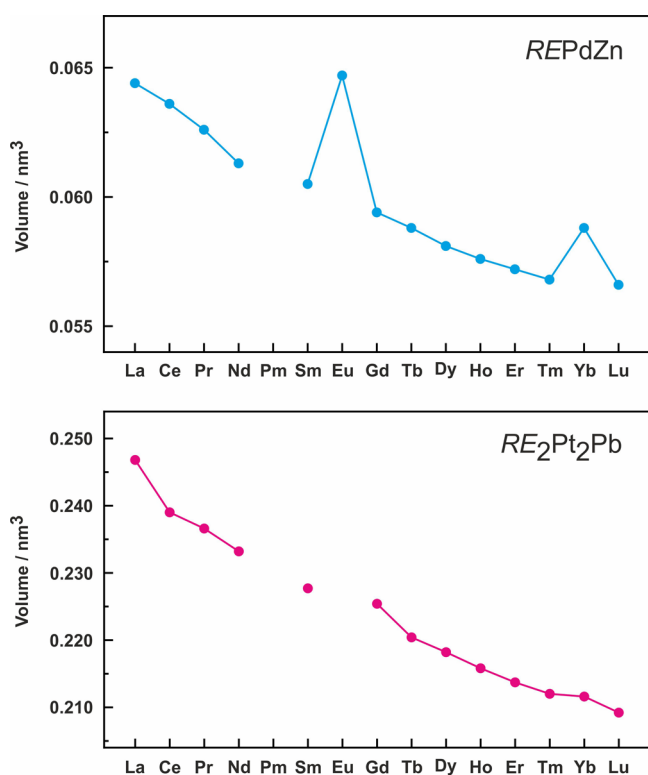
## CRYSTAL CHEMICAL PRINCIPLES

The present review has a higher focus on the physical properties of ytterbium intermetallics, and we therefore only give a short and concise overview on some of the basic crystal chemical principles that are important for the family of  $\text{Yb}_x\text{T}_y\text{X}_z$  intermetallics. The structural chemistry is mostly driven by the ytterbium valence. As discussed above, the filled 4f shell of Yb(II) might be considered as a diamagnetic core–electron shell, and consequently, we observe a crystal chemical behavior of an alkaline earth element. This is readily understandable when comparing the ionic radii for coordination number 6 for  $\text{Ca}^{2+}$  (100 pm) and  $\text{Yb}^{2+}$  (102 pm).<sup>43</sup> We thus observe a close structural relationship of the divalent ytterbium intermetallics with their respective calcium analogues. Typical examples are the binary Zintl phases  $\text{Ca}_2\text{Ge}$ <sup>44</sup> and  $\text{Yb}_2\text{Ge}$ <sup>45</sup> as well as the ternary Zintl phases  $\text{CaZnSn}$ <sup>46</sup> and  $\text{YbZnSn}$ <sup>47</sup> with many more pairs of isotypic compounds than for  $\text{Yb}^{2+}/\text{Ca}^{2+}$ .<sup>48</sup>

A similar close relationship is known for  $\text{Eu}^{2+}$  (117 pm) and  $\text{Sr}^{2+}$  (118 pm).<sup>43</sup> Some examples are  $\text{EuAl}_2\text{Pt}$  and  $\text{SrAl}_2\text{Pt}$  ( $\text{MgCuAl}_2$  type, space group  $Cmcm$ ),<sup>49</sup>  $\text{EuAl}_3\text{Pt}_3$ <sup>50</sup> and  $\text{SrAl}_3\text{Pt}_3$ <sup>51</sup> ( $\text{YNi}_5\text{Si}_3$  type, space group  $Pnma$ ),  $\text{SrAu}_3\text{Al}_2$ <sup>52</sup> and  $\text{EuAu}_3\text{Al}_2$ <sup>53</sup> ( $\text{RbAu}_3\text{Ga}_2$  type, space group  $Pnma$ ),  $\text{EuAu}_2\text{Al}_2$ <sup>54</sup> and  $\text{SrAu}_2\text{Al}_2$ <sup>54,55</sup> ( $\text{CaBe}_2\text{Ge}_2$  type, space group  $P4/nmm$ ),  $\text{Eu}_2\text{T}_2\text{In}$ <sup>56</sup> and  $\text{Sr}_2\text{T}_2\text{In}$ <sup>57</sup> ( $T = \text{Pd}, \text{Pt}; \text{Ca}_2\text{Ir}_2\text{Si}$  type, space

group  $C2/c$ ),  $\text{EuRh}_2\text{In}_8$ <sup>58</sup> and  $\text{SrRh}_2\text{In}_8$ <sup>59</sup> ( $\text{CeFe}_2\text{Al}_8$  type, space group  $Pbam$ ),  $\text{EuPtSi}$  and  $\text{SrPtSi}$  ( $\text{LaIrSi}$  type, space group  $Pnma$ ),<sup>60</sup> or finally  $\text{EuPt}_2\text{Si}_2$  and  $\text{SrPt}_2\text{Si}_2$  ( $\text{ThCr}_2\text{Si}_2$  type, space group  $I4/mmm$ ).<sup>61</sup> Besides the direct comparisons  $\text{Sr}^{2+}/\text{Eu}^{2+}$  and  $\text{Ca}^{2+}/\text{Yb}^{2+}$ , it is not always possible to directly compare an  $\text{Eu}(\text{II})$  example and an  $\text{Yb}(\text{II})$  example within one series of compounds, since (i) the structure type is not formed with both cations or (ii) the given ytterbium is trivalent. And even if both compounds are known and adopt the same structure type, the physical properties are significantly different due to  $\text{Yb}(\text{II})$  being a diamagnetic cation while  $\text{Eu}(\text{II})$  exhibits a half-filled 4f shell ( $4f^7$ ).

In the case of trivalent ytterbium intermetallics, the ytterbium compound usually adopts the crystal structure of the neighboring compounds existing with thulium or lutetium. A first fast hint for the ytterbium valence can be gathered from the plot of the unit cell volumes of a given series of rare earth compounds, a so-called Iandelli plot. As examples, we present the series of  $\text{REPdZn}$ <sup>62–64</sup> and  $\text{RE}_2\text{Pt}_2\text{Pb}$ <sup>47,65</sup> phases in Figure 1. Since divalent ytterbium has a distinctly larger radius than

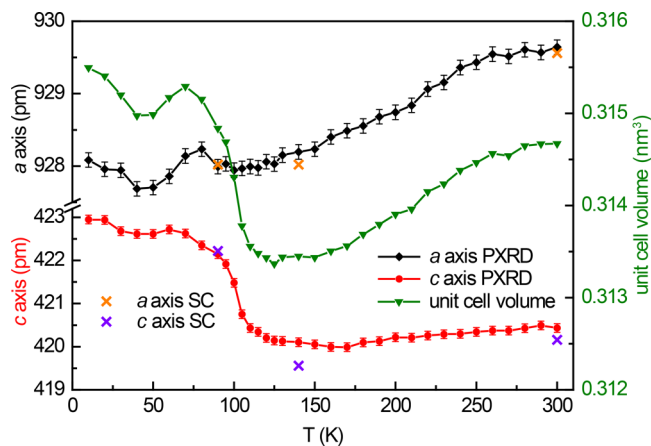


**Figure 1.** Plot of the unit cell volumes for the series  $\text{REPdZn}$  (per formula unit) and  $\text{RE}_2\text{Pt}_2\text{Pb}$  (subcell volumes). Data was taken from the Pearson database.<sup>18</sup>

trivalent ytterbium, positive deviations in the Iandelli plot (this is the case for the  $\text{REPdZn}$  series) indicate divalent ytterbium. If the unit cell volume fits smoothly into the plot, then ytterbium is trivalent (this is the case for the  $\text{RE}_2\text{Pt}_2\text{Pb}$  series). Nevertheless, one important observation needs to be mentioned at this point. Although the physical property measurements (vide infra) show stable trivalent ytterbium in  $\text{Yb}_2\text{Pt}_2\text{Pb}$ , the Iandelli plot shows a very small positive deviation. Such an anomaly is frequently observed for rare earth series with trivalent ytterbium. Another striking series with such a weak positive deviation is the one of  $\text{REAL}_3\text{C}_3$ .<sup>66</sup>

Many more examples for both cases can be found when checking structural databases like the Pearson database.<sup>18</sup>

Compounds that exhibit temperature- or pressure-dependent valence phase transitions show a discontinuity in their lattice parameters.  $\text{YbPd}_2\text{Al}_3$  ( $\text{YNi}_2\text{Al}_3$  type, space group  $P6/mmm$ ), for example, exhibits a mixed valent state ( $\text{Yb}^{\text{II}}$  and  $\text{Yb}^{\text{III}}$ ) at room temperature which transitions into an almost fully divalent state below  $T_{\text{trans}} = 110$  K. The temperature dependence of the lattice parameters is shown in Figure 2. Here, a clear increase of the  $c$  lattice parameter is



**Figure 2.** Lattice parameters of  $\text{YbPd}_2\text{Al}_3$  obtained from temperature-dependent powder X-ray diffraction experiments upon cooling from 300 to 10 K and from single-crystal X-ray diffraction experiments. Reprinted with permission from ref 67.

visible (+3 pm, 0.7%). Due to the relatively high transition temperature, single crystal studies were possible, that confirmed the thermal behavior of the lattice parameters and clearly indicated that no structural phase transition takes place. However, significant distortions in the respective Yb coordination polyhedra are observed.<sup>67</sup>

Finally, a very recent study of  $\text{Yb}_2\text{Pd}_2\text{Cd}$  should be mentioned. Here, a structural phase transition at  $\sim 150$  K was observed. The high-temperature modification adopts the  $\text{Mo}_2\text{B}_2\text{Fe}$  type (space group  $P4/mbm$ ) while below the phase transition temperature, LT- $\text{Yb}_2\text{Pd}_2\text{Cd}$  is found in a new structure type (space group  $P4/mbm$ ) with a doubled  $c$  axis. The striking structural feature concerns the ytterbium–palladium coordination. The average Yb–Pd distances distort, forming a smaller and a larger coordination environment and enabling a charge separation for  $\text{Yb}^{2+}$  and  $\text{Yb}^{3+}$ .<sup>68</sup>

## SELECTED CRYSTAL STRUCTURES

In the present chapter, we want to selectively focus on some structural principles and refer to extended crystal chemical review articles for further reading. Some of the  $\text{Yb}_x\text{T}_y\text{X}_z$  phases crystallize with their individual structure type which have all been thoroughly described in the original articles.

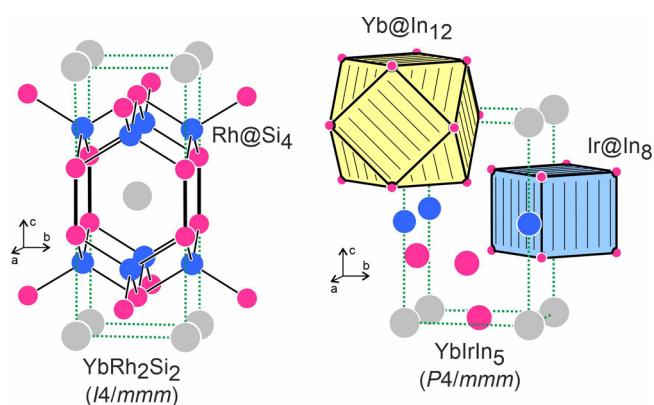
Many of the intermetallic ytterbium compounds belong to the so-called equiatomic phases  $\text{YbTX}$  ( $T$  = (transition) metal;  $X$  = element of group 13, 14, or 15). The most frequent structure types that occur for these phases are  $\text{TiNiSi}$  (space group  $Pnma$ ),  $\text{ZrNiAl}$  (space group  $P62m$ ),  $\text{MgAgAs}$  (space group  $F43m$ ),  $\text{PbFCl}$  (space group  $P4/nmm$ ), and other  $\text{AlB}_2$  (space group  $P6/mmm$ ) derived superstructures. Their crystal chemistry has been reviewed in the *Handbook on the Physics*

and Chemistry of Rare Earths.<sup>13</sup> These structure types allow for a certain structural flexibility. To give an example, for the orthorhombic TiNiSi type, a manifold of ytterbium-based phases YbTX ( $T = \text{Mg, Ni, Cu, Rh, Pd, Pt, Au}$ ;  $X = \text{Al, Si, P, Ga, Ge, Sn, Sb}$ ) have been synthesized.<sup>13,18</sup> Apart from the overview on the equiatomic YbTX phases, the crystal chemistry of these versatile structure types has repeatedly been discussed in various other review articles.<sup>3–6,8,69–72</sup> We refer to these overviews for further reading.

A huge number of property studies has been conducted for the binary aluminides YbAl<sub>2</sub> (MgCu<sub>2</sub> type, cubic Laves phase, space group  $Fd\bar{3}m$ )<sup>73</sup> and YbAl<sub>3</sub> (Cu<sub>3</sub>Au type; ordered cubic closest packing, space group  $Pm\bar{3}m$ ).<sup>74,75</sup> Both compounds crystallize with well-known basic structure types. In this context it is interesting to note that ternary ones have also been reported with the structure types of the cubic and hexagonal Laves phases. In the case of the MgCu<sub>2</sub> type phases, especially the solid solution members Yb<sub>1-x</sub>In<sub>x</sub>Cu<sub>2</sub> have been studied, where the ytterbium substructure has been magnetically diluted through indium substitution. YbCu<sub>4</sub>In<sup>76</sup> shows 1:1 Yb/In ordering and crystallizes with the noncentrosymmetric MgCu<sub>4</sub>Sn type (space group  $F43m$ ). The crystal chemical details of Laves phase superstructures have recently been reviewed.<sup>77</sup> Several aluminides were reported as equiatomic compounds, and the MgZn<sub>2</sub> type (space group  $P6_3/mmc$ ) was assigned.<sup>13</sup> Meanwhile, it is known that an equiatomic composition with complete atom ordering is not possible. The superstructure formation leads to a small deviation. The correct model was solved for Yb<sub>6</sub>Ir<sub>5</sub>Ga<sub>7</sub> (space group  $P6_3/mcm$ ) and the series of gallides RE<sub>6</sub>Ir<sub>5</sub>Ga<sub>7</sub>.<sup>78,79</sup> Shortly later, also the aluminum containing RE<sub>6</sub>T<sub>5</sub>Al<sub>7</sub> phases ( $RE = \text{Sc, Y, Ce–Nd, Sm, Gd–Lu}$ ;  $T = \text{Ru, Ir}$ )<sup>80</sup> have been structurally characterized. With respect to the physical properties, one must differentiate between the fully ordered 6–5–7 phases and solid solutions where mixed occupied sites occur. Often, disorder destroys magnetic ordering or drastically reduces the magnetic ordering temperature.

We now discuss the crystal chemistry of some selected examples. One of the striking ternary ytterbium intermetallics is YbRh<sub>2</sub>Si<sub>2</sub> (ThCr<sub>2</sub>Si<sub>2</sub> type, space group  $I4/mmm$ ) originally reported in 1979 by Rossi et al.<sup>22</sup> Due to its outstanding physical properties (vide infra), meanwhile >350 publications are listed in SciFinder.<sup>19</sup> The YbRh<sub>2</sub>Si<sub>2</sub> unit cell is presented in Figure 3. The rhodium atoms have a tetrahedral silicon coordination with Rh–Si distances of 235 pm. These [RhSi<sub>4</sub>] tetrahedra are condensed via edge-sharing, building a dense layer in the  $ab$  plane. Adjacent layers are connected via Si–Si bonds (246 pm). The resulting network provides large cages for the ytterbium atoms which have a coordination number of 18, i.e., Yb@Si<sub>10</sub>Rh<sub>8</sub>. Due to the large cages, the shortest Yb–Yb distances are 401 pm, corresponding to the  $a$  lattice parameter. The ThCr<sub>2</sub>Si<sub>2</sub> family of compounds is one of the largest among all intermetallic compounds with more than 4000 entries in the Pearson database.<sup>18</sup> The crystal chemical details for these materials are summarized in diverse overviews.<sup>1,69,81,82</sup>

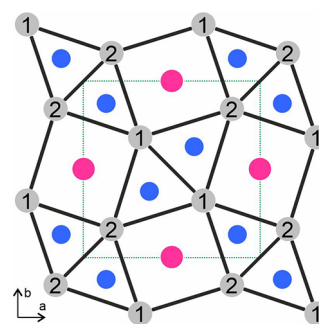
Another, smaller series of tetragonal compounds concerns those with HoCoGa<sub>5</sub> type structure<sup>84</sup> (space group  $P4/mmm$ ). This type has been observed for YbCoGa<sub>5</sub> and the indides YbTIn<sub>5</sub> ( $T = \text{Co, Rh, Ir}$ ).<sup>83,85</sup> The YbIrIn<sub>5</sub> unit cell<sup>83</sup> is also presented in Figure 3. The YbIrIn<sub>5</sub> structure is composed of two striking building units. The ytterbium atoms have slightly tetragonally distorted cuboctahedral indium coordination ( $8 \times$



**Figure 3.** Crystal structures of YbRh<sub>2</sub>Si<sub>2</sub><sup>22</sup> and YbIrIn<sub>5</sub>.<sup>83</sup> Ytterbium, rhodium (iridium), and silicon (indium) atoms are drawn as medium gray, blue, and magenta circles, respectively. The network of edge-sharing [RhSi<sub>4</sub>] tetrahedra for YbRh<sub>2</sub>Si<sub>2</sub> and the characteristic polyhedra for YbIrIn<sub>5</sub> are emphasized.

320 and  $4 \times 327$  pm Yb–In), which can be considered as a cutout of an ordered closest  $fcc$  related packing (given the comparable size of ytterbium and indium). These cuboctahedra condense to layers via four common rectangular faces. These layers alternate with layers of condensed Ir@In<sub>8</sub> square prisms (275 pm Ir–In). Similar to YbRh<sub>2</sub>Si<sub>2</sub> discussed above, the ytterbium atoms in YbIrIn<sub>5</sub> are also well separated from each other. The shortest Yb–Yb distance of 462 pm again corresponds to the lattice parameter  $a$ .

Yb<sub>2</sub>Pt<sub>2</sub>Pb<sup>43</sup> is one of the few ytterbium intermetallics that belong to the family of U<sub>3</sub>Si<sub>2</sub> derived phases. Due to a puckering effect, one observes superstructure formation for Yb<sub>2</sub>Pt<sub>2</sub>Pb (space group  $P4_2/mmm$ ) with a small distortion of the ytterbium trigonal and square prisms. For better visibility, only one layer is presented in Figure 4. A more detailed crystal

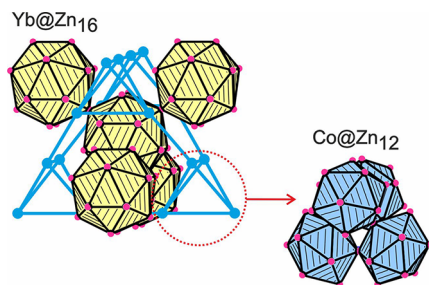


**Figure 4.** Cutout of the Yb<sub>2</sub>Pt<sub>2</sub>Pb<sup>47</sup> structure. Ytterbium, platinum, and lead atoms are drawn as medium gray, blue, and magenta circles, respectively. The CsCl and AlB<sub>2</sub> related slabs and the two crystallographically independent ytterbium sites are emphasized. Only one layer is shown for better visibility.

chemical discussion is given in references 86 and 87. The striking feature of this plumbide is the Shastry–Sutherland-like substructure of the ytterbium atoms, which gives rise to magnetic frustration (discussed in detail in the magnetic part, vide infra). Several years after the discovery of the compound,<sup>47</sup> this has motivated many detailed property studies, which are addressed below.

A complex cubic structure occurs for the aluminum- and zinc-rich compounds YbT<sub>2</sub>Al<sub>20</sub> ( $T = \text{Ti, V, Cr}$ ) and YbT<sub>2</sub>Zn<sub>20</sub> ( $T = \text{Co, Rh, Ir}$ ).<sup>18</sup> They crystallize with the CeCr<sub>2</sub>Al<sub>20</sub> type,

space group  $Fd\bar{3}m$  with 184 atoms per unit cell. The main interest in these ytterbium compounds concerned their heavy-Fermion type behavior. The many facets of their physical properties have recently been summarized.<sup>88</sup> As an example we present the  $\text{YbCo}_2\text{Zn}_{20}$  structure<sup>89</sup> in Figure 5. Although the

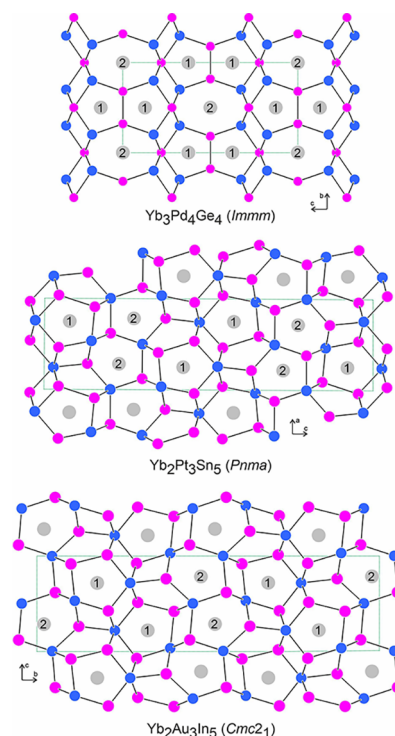


**Figure 5.** Cutout of the  $\text{YbCo}_2\text{Zn}_{20}$ <sup>89</sup> structure. Cobalt and zinc atoms are drawn as medium blue and magenta circles, respectively. The connectivity pattern of the  $\text{Yb@Zn}_{16}$  and  $\text{Co@Zn}_{12}$  polyhedra is emphasized. For details see text.

structure seems complex at first sight, it can easily be constructed from two basic building units. The ytterbium and cobalt atoms of  $\text{YbCo}_2\text{Zn}_{20}$  form a substructure that is isopointal with the cubic Laves phase  $\text{MgCu}_2$ , i.e., has the same space group symmetry. The zinc atoms now surround the ytterbium and cobalt atoms, leading to  $\text{Yb@Zn}_{16}$  and  $\text{Co@Zn}_{12}$  polyhedra which condense in a tetrahedral motif. Again, the ytterbium atoms are well separated from each other and show no bonding contacts. This “highly diluted” rare earth substructure is prone for basic property studies.

In Figure 6, we now turn to a series of ytterbium compounds with complex extended polyanionic networks which exhibit multiple ytterbium sites. Exemplarily, we present the structures of  $\text{Yb}_3\text{Pd}_4\text{Ge}_4$ ,<sup>90</sup>  $\text{Yb}_2\text{Pt}_3\text{Sn}_5$ ,<sup>29</sup> and  $\text{Yb}_2\text{Au}_3\text{In}_5$ .<sup>91</sup> The common structural feature concerns the dense condensation of the transition metal and p element atoms within the polyanionic networks  $[\text{Pd}_4\text{Ge}_4]$ ,  $[\text{Pt}_3\text{Sn}_5]$ , and  $[\text{Au}_3\text{In}_5]$ . The latter are stabilized through Pd–Ge, Pt–Sn, Au–In as well as Ge–Ge, Sn–Sn, and In–In bonding interactions. The networks then leave cavities of different sizes which are filled by the ytterbium atoms. This leads to the coordination polyhedra  $\text{Yb1@Ge}_6\text{Pd}_6$  and  $\text{Yb2@Ge}_6\text{Pd}_8$  for  $\text{Yb}_3\text{Pd}_4\text{Ge}_4$ ,  $\text{Yb1@Sn}_6\text{Pt}_2$  and  $\text{Yb2@Pt}_5\text{Sn}_8$  for  $\text{Yb}_2\text{Pt}_3\text{Sn}_5$  as well as  $\text{Yb1@Au}_4\text{In}_{10}$  and  $\text{Yb2@Au}_5\text{In}_8$  for  $\text{Yb}_2\text{Au}_3\text{In}_5$ . The three compounds tend to exhibit mixed ytterbium valences. From a crystal chemical point of view, the smaller cavities are filled by trivalent ytterbium (static mixed valency). For  $\text{Yb}_2\text{Pt}_3\text{Sn}_5$ , this feature was tested also from a synthetic point of view. Given the comparable size of  $\text{Yb}^{2+}$  and  $\text{Ca}^{2+}$  it was possible to synthesize the stannides  $\text{CaErPt}_3\text{Sn}_5$ ,  $\text{CaTmPt}_3\text{Sn}_5$ ,  $\text{CaYbPt}_3\text{Sn}_5$ , and  $\text{CaLuPt}_3\text{Sn}_5$  with calcium site preference on the “divalent” place.<sup>92</sup>

The last crystal chemical example concerns ytterbium aluminum borides. Figure 7 shows projections of the  $\alpha$ - and  $\beta$ - $\text{YbAlB}_4$ <sup>93</sup> and  $\text{Yb}_2\text{AlB}_6$ <sup>94</sup> structures along the short unit cell axis. The boron atoms build up planar layers with different tessellations, which are stacked in a ...AA... sequence. Between the layers, pentagonal prisms for the smaller aluminum atoms and hexagonal, respectively, heptagonal ones for the larger ytterbium atoms are formed. The two modifications of  $\text{YbAlB}_4$  show ytterbium exclusively in the heptagonal prisms, while two prism types occur in  $\text{Yb}_2\text{AlB}_6$ . The  $\text{YbAlB}_4$  modifications gained high interest in recent years since they exhibit heavy



**Figure 6.** View of the  $\text{Yb}_3\text{Pd}_4\text{Ge}_4$ ,<sup>90</sup>  $\text{Yb}_2\text{Pt}_3\text{Sn}_5$ ,<sup>29</sup> and  $\text{Yb}_2\text{Au}_3\text{In}_5$ <sup>91</sup> structures along the short unit cell axis. Ytterbium, palladium (platinum, gold), and germanium (tin, indium) atoms are drawn as medium gray, blue, and magenta circles, respectively. The  $[\text{Pd}_4\text{Ge}_4]$ ,  $[\text{Pt}_3\text{Sn}_5]$ , and  $[\text{Au}_3\text{In}_5]$  networks and the crystallographically independent ytterbium sites are emphasized.

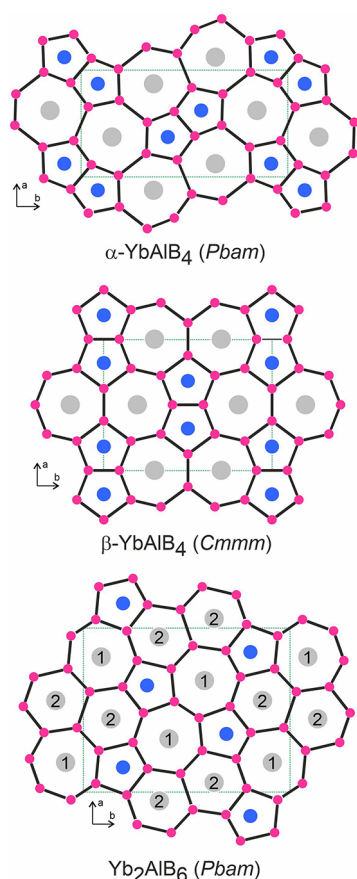
Fermion ground state and superconductivity at low temperatures (vide infra). Already more than 120 publications appeared for this composition.<sup>19</sup>

## PHYSICAL PROPERTIES AND SPECTROSCOPY

Given the large number of known phases and the fact that some of them have only been characterized with respect to their structure, the review focuses on the ytterbium valence and selected physical properties that arise from these. Thus, selected compounds with a divalent or trivalent ytterbium state are discussed to outline these properties.

Most compounds have routinely been studied with respect to their magnetic behavior in order to get a first hint for the ytterbium valence. A significantly smaller section of these studies is additionally accompanied by resistivity and specific heat measurements; however, these measurements are usually not routinely conducted. Therefore, magnetic, resistivity, and specific heat investigations are, where applicable, discussed together in the following chapter. Subsequently, separate chapters summarize the data on more sophisticated investigation techniques such as (i) inelastic neutron scattering, (ii) XANES/XAS, (iii) X-ray photoelectron spectroscopy (XPS), (iv) nuclear magnetic and electron resonance spectroscopy (NMR and ESR), (v)  $^{171}\text{Yb}$  Mössbauer spectroscopy, and (vi) thermoelectric measurements.

Within each property chapter, the data has been summarized with respect to composition (binary or ternary) as well as structure types or structural families (e.g., Zintl phases), facilitating the comparability with respect to the valence electron count, etc.



**Figure 7.** View of the  $\alpha$ -YbAlB<sub>4</sub>,<sup>93</sup>  $\beta$ -YbAlB<sub>4</sub>,<sup>93</sup> and Yb<sub>2</sub>AlB<sub>6</sub>,<sup>94</sup> structures along the short unit cell axis. Ytterbium, aluminum, and boron atoms are drawn as medium gray, blue, and magenta circles, respectively. The boron nets and the crystallographically independent ytterbium sites are emphasized.

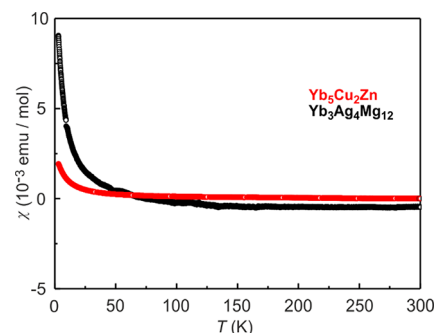
### MAGNETISM, RESISTIVITY AND SPECIFIC HEAT

Before starting with a broader discussion on the magnetic properties of intermetallic ytterbium compounds, we need to comment on ytterbium sesquioxide, Yb<sub>2</sub>O<sub>3</sub>, the most inconvenient byproduct occurring during the synthesis of Yb intermetallics. Yb<sub>2</sub>O<sub>3</sub> exhibits paramagnetism (4f<sup>13</sup> configuration;  $T_N = 2.3$  K<sup>95</sup>) and can dominate the susceptibility in the low-temperature regime,<sup>96</sup> especially in the case of intermediate-valent or almost divalent ytterbium intermetallics. In those cases, a clear quantification of the Yb<sub>2</sub>O<sub>3</sub> content is necessary along with a correction procedure.<sup>97</sup> If this feature is not taken into account, the interpretation of the magnetic data remains ambiguous. Since many of the ytterbium intermetallics order magnetically at very low temperature, the Yb<sub>2</sub>O<sub>3</sub> impurity might overlay the intrinsic magnetism of the sample.

#### Divalent Ytterbium Compounds

Purely divalent ytterbium compounds have the valence electron configuration 4f<sup>14</sup>, which exhibits high stability. This filled 4f shell can be considered as a core-electron shell; therefore, no magnetic moment is observed. Such ytterbium intermetallics behave either as diamagnets or as Pauli paramagnets. Since most of these phases are metallic conductors, the observed susceptibility usually arises from an overcompensation of the intrinsic core diamagnetism by the Pauli susceptibility caused by the conduction electrons.

Exemplarily, the temperature dependences of the magnetic susceptibility of Yb<sub>5</sub>Cu<sub>2</sub>Zn and Yb<sub>3</sub>Ag<sub>4</sub>Mg<sub>12</sub> are given in Figure 8. Above 100 K, the susceptibilities are very weak and almost



**Figure 8.** Temperature dependence of the magnetic susceptibility  $\chi$  of Yb<sub>3</sub>Ag<sub>4</sub>Mg<sub>12</sub><sup>99</sup> and Yb<sub>5</sub>Cu<sub>2</sub>Zn,<sup>98</sup> measured at 10 kOe.

temperature-independent. The room temperature value of  $7.2 \times 10^{-6}$  emu mol<sup>-1</sup> for Yb<sub>5</sub>Cu<sub>2</sub>Zn indicates Pauli paramagnetism.<sup>98</sup> In contrast, in Yb<sub>3</sub>Ag<sub>4</sub>Mg<sub>12</sub> the intrinsic diamagnetism is not as strongly overcompensated by the Pauli contribution, leading to a room temperature value of  $-4.61 \times 10^{-4}$  emu mol<sup>-1</sup>.<sup>99</sup> The susceptibility increases in the low-temperature regimes (the so-called Curie tails), which originates from traces of paramagnetic impurities. Further examples of room temperature susceptibilities are given in Table 1. In these compounds, the ytterbium atoms are divalent and therefore carry no magnetic moment.

**Table 1.** Magnetic Susceptibilities of Selected Ytterbium Intermetallics at Room Temperature<sup>a</sup>

Compound	$\chi$ /emu mol <sup>-1</sup>	Reference
YbAgMg	$+1.2(1) \times 10^{-4}$	100
YbAuMg	$+2.4 \times 10^{-4}$	101
Yb <sub>3</sub> Ag <sub>4</sub> Mg <sub>12</sub>	$-4.61 \times 10^{-4}$	99
Yb <sub>5</sub> Cu <sub>2</sub> Zn	$+7.2 \times 10^{-6}$	98
YbAgCd	$+4.8 \times 10^{-5}$	102
YbAuCd	$+5.8 \times 10^{-5}$	101
YbPdIn <sub>2</sub>	$-4.9(2) \times 10^{-4}$	103
YbCuSn	$+4.22 \times 10^{-4}$	104
YbAgSn	$-1.37 \times 10^{-5}$	104
YbAuSn	$-2.62 \times 10^{-5}$	104
YbZnSn	$+2.0(1) \times 10^{-4}$	47
YbCuSb	$+1.38 \times 10^{-4}$	104
YbAgSb	$-7.69 \times 10^{-5}$	104
YbAuSb	$-8.47 \times 10^{-5}$	104

<sup>a</sup>Standard deviations are given in parentheses, where available.

#### Trivalent Ytterbium Compounds

Trivalent ytterbium cations have a 4f<sup>13</sup> valence electron configuration. The  $g$  value of 8/7 and the total angular momentum of  $J = 7/2$  lead to a magnetic moment of  $\mu = \sqrt{g_J[J(J+1)]} = 4.54 \mu_B$  per Yb<sup>3+</sup>.<sup>105</sup> The f electrons may play an important role in bonding/hybridization of intermetallic ytterbium compounds, leading to interesting properties, such as heavy Fermion behavior, Kondo behavior, intermediate valence, or complex magnetic structures. These properties can be influenced by different parameters, like the chemical environment, the magnetic field, temperature, or pressure.<sup>106</sup>

Table 2. Magnetic Properties of Selected Ytterbium Compounds<sup>a</sup>

Compound	$T_C$ /K	$T_N$ /K	$\mu_{\text{eff}}/\mu_B$	$\gamma$ /mJ mol <sup>-1</sup> K <sup>-2</sup>	Reference
YbAlB <sub>4</sub>				125	204,222
YbAl <sub>3</sub> C <sub>3</sub>			4.66		177
YbFe <sub>2</sub> Zn <sub>20</sub>			Near 4.53	520	88,183–186
YbCo <sub>2</sub> Zn <sub>20</sub>			Near 4.53	7900	88,183–186
YbNiAl <sub>4</sub>			2.9	16	23,215,216
YbNi <sub>3</sub> Al <sub>9</sub>	–	3.4	4.37	110	223
YbNi <sub>4</sub> Si			4.15	25	224
YbNiGa		1.7, 1.9	4.4	450	126
YbNiSn	5.65	–	4.3	300	121–123,125
YbCuAl			4.32	260	146,147,225
YbCu <sub>2</sub> Si <sub>2</sub>			4.19	135	226
YbPdCu <sub>4</sub>	–	0.8	4.33	200	227
YbInCu <sub>4</sub>			4.37	50–400	228,229
YbRh <sub>2</sub> Si <sub>2</sub>	–	0.07	$2 \times 10^{-3}$		155,161
YbRhSn	–	1.4, 1.85	4.3–4.5	1200(200)	110,126–128
YbRhSb	2.7	–	1.4	370	129–131
Yb <sub>4</sub> Rh <sub>7</sub> Ge <sub>6</sub>			4.55–4.65	0.184	230
YbPdBi	1	–	4.11(5)	1200(100)	110,127
Yb <sub>4</sub> Ir <sub>7</sub> Ge <sub>6</sub>			4.8	0.281	230
YbPtGe	5.4	–	4.48	209	231
YbPtIn		1.4, 3.4	4.21–4.4	400	126,232
YbPtSn	3.5(4)	–	4.27(2)		127
Yb <sub>2</sub> Pt <sub>2</sub> Pb	–	2.07	4.42–4.54	30	173,233
YbPtBi	–	0.4		8000	135–139

<sup>a</sup>Magnetic properties include Néel temperature,  $T_N$ ; Curie temperature,  $T_C$ ; effective magnetic moment,  $\mu_{\text{eff}}$  and specific heat capacity (Sommerfeld) coefficient ( $\gamma$ ). Standard deviations are given in parentheses, where available.

Characteristic for heavy Fermion behavior is the large electronic specific heat coefficient  $\gamma$  at low temperatures, 2 or 3 orders of magnitude larger (arbitrary value of 400 mJ mol<sup>-1</sup> K<sup>-2</sup> according to ref 107) than that of copper ( $\sim 1$  mJ mol<sup>-1</sup> K<sup>-2</sup>).<sup>108</sup> In addition, heavy Fermion compounds typically have large Pauli susceptibilities at low temperatures, due to the large density of states.<sup>109</sup> It is very difficult to establish the heavy Fermion behavior of ytterbium compounds, due to the challenging distinction between magnetic, crystal field, and Kondo energy scales. Thus, many compounds show both, magnetic and heavy Fermion character.<sup>110</sup>

The common way to fit the dependence of the magnetic susceptibility is using the Curie–Weiss law. Within the modified Curie–Weiss law, contributions of the conduction electrons are also taken into account. This makes it possible to adjust even slight curvatures of the reciprocal susceptibility. However, both laws do not consider the temperature dependency of the effective magnetic moment, which marks the only way to explain decrease of the  $\chi^{-1}$  value. The ICF model (interconfiguration fluctuation model),<sup>111</sup> developed by Sales and Wohleben, addresses a possible mixed-valent behavior of rare earth atoms (like Yb or Ce) and is represented by

$$\chi_{\text{ICF}}(T) = \frac{N\mu_{\text{eff}}^2[1 - \nu(T)]}{3k_B(T + T_{\text{sf}})}$$

with  $T_{\text{sf}}$  as the spin fluctuation temperature and  $\nu(T)$  as the temperature-dependent mean occupation of the ground state given by

$$\nu(T) = \frac{1}{1 + 8\exp[-E_{\text{ex}}/k_B(T + T_{\text{sf}})]}$$

where  $E_{\text{ex}}$  is the energy gap between the ground and the excited state. In most cases, the experimental  $\chi(T)$  data can be described by

$$\chi(T) = \chi_{\text{ICF}}(T) + \chi_{\text{CW}}(T) + \chi_0$$

consisting in part of the ICF model ( $\chi_{\text{ICF}}$ ), the Curie–Weiss impurity ( $\chi_{\text{CW}}$ ), and  $\chi_0$ , the latter being a sum of temperature-independent contributions.<sup>111,112</sup>

In the following paragraph, some striking examples will be briefly discussed. The magnetic properties and specific heat coefficients of these and further compounds were summarized in Table 2.

### Binary Compounds

During the last 50 years, YbAl<sub>2</sub> (MgCu<sub>2</sub> type, space group  $Fd\bar{3}m$ ) and YbAl<sub>3</sub> (Cu<sub>3</sub>Au type, space group  $Pm\bar{3}m$ ) have been extensively examined regarding their physical properties. Both compounds show Curie–Weiss behavior above 100 K, with a broad maximum of the susceptibility around 850 K for YbAl<sub>2</sub>. The detected characteristic spin fluctuation temperatures are  $T_{\text{sf}} = 200$  K for YbAl<sub>2</sub> and  $T_{\text{sf}} = 125$  K in case of YbAl<sub>3</sub>.<sup>13,114</sup> In a hard X-ray photoemission spectroscopy study for YbAl<sub>2</sub>, below 300 K, a ytterbium valence of ca. +2.2 was determined.<sup>115,116</sup> The Kondo temperature suggested from the magnetic measurements is very high ( $T_K > 2000$  K).<sup>97,117</sup> The valence of YbAl<sub>3</sub> (+2.75) results in a ground state with 4f occupancy of 0.75.<sup>118,119</sup> In contrast, photoemission experiments revealed a valence of +2.73, whereas the determined Kondo temperature is in an order of 500–600 K, and the specific heat coefficient is  $\gamma = 40$  mJ/mol K<sup>2</sup>. Below the Fermi-liquid temperature  $T_{\text{fl}} = 50$  K, a quadratic temperature dependence of the resistivity was found.<sup>118–120</sup>

### Equiatomic Ternary Compounds

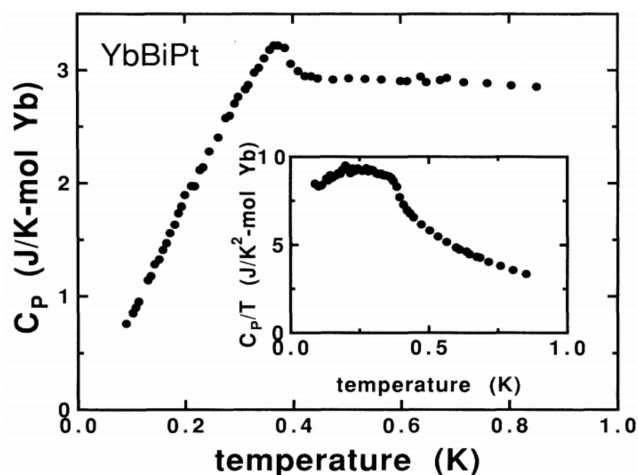
Weak ferromagnetism is reported for the equiatomic compound YbNiSn (TiNiSi type, space group  $Pnma$ ). The dense Kondo compound has a Curie-temperature of  $T_C = 5.65$  K, and the magnetic moment lies along the  $c$  axis. A value of  $\mu_{\text{eff}} = 4.3 \mu_B$  was determined as an effective magnetic moment close to the value of a free Yb<sup>3+</sup> ion ( $4.54 \mu_B$ ). The maximum detected magnetization is  $0.85 \mu_B$  per Yb atom. Kondo fluctuations with a Kondo temperature  $T_K = 2$  K lead to the reduction of the ferromagnetism.<sup>121–124</sup> Further investigations by Mössbauer spectroscopy, neutron diffraction, and resistivity measurements confirm these results.<sup>121–125</sup>

The susceptibility of YbRhSn with ZrNiAl type structure (space group  $P62m$ ) follows the Curie–Weiss law at high temperatures, with a trivalent state of the Yb ions. The deduced effective magnetic moment was in a range of 4.4–4.5  $\mu_B$ , and a negative Weiss constant was observed.<sup>126–128</sup> Two different magnetic transitions at  $T = 1.85$  and 1.4 K were reported via heat capacity data, which shift to lower temperatures with applied magnetic fields as expected for antiferromagnetic ordering. In addition, the heat capacity data indicate a possible heavy Fermion ground state. The resistivity of YbRhSn is Kondo-like and decreases almost linearly down to 20 K, followed by a rapid decrease at 1.8 K. Below 1 K it has a Fermi-liquid-like temperature variation.<sup>126–128</sup>

YbRhSb was first synthesized by Muro et al. in 2004 and crystallizes with the TiNiSi type structure. It reveals a ferromagnetic transition at  $T_C = 2.7$  K. For an applied magnetic field parallel to the crystallographic  $b$  axis, a spontaneous moment of only  $\sim 3.0 \times 10^{-3} \mu_B$  per Yb atom was observed, although the magnetic easy axis is the  $a$  axis. At around 2 T a metamagnetic transition occurs, and at 15 T the magnetization reaches  $1.4 \mu_B/\text{Yb}$  for the  $B\parallel a$  axis, somewhat reduced when compared with the theoretical saturation magnetization of  $4 \mu_B$ .<sup>105</sup> The specific heat coefficient was determined to be  $\gamma \sim 370 \text{ mJ mol}^{-1} \text{ K}^{-2}$  in the ferromagnetically ordered state.<sup>129</sup> Instead of typical ferromagnets, the Curie temperature furthermore decreases with higher applied magnetic fields. Thus, the weak ferromagnetism of YbRhSb is possibly the result of a canted antiferromagnetic structure. Resistivity measurements and investigations in a  $p/T$  phase diagram revealed a minimum for  $T_M(p)$  at  $p_C = 1.7$  GPa. For  $p \geq 2$  GPa, the canted antiferromagnetic structure changes to a ferromagnetic structure and a moment of  $0.4 \mu_B/\text{Yb}$  within the  $bc$  plane.<sup>130–133</sup>

YbBiPt crystallizes in the cubic MgAgAs type structure with space group  $F\bar{4}3m$ .<sup>134</sup> Antiferromagnetic ordering was detected at  $T_N = 0.4$  K, while the determined Weiss constant is  $\Theta_p = 0.4$  K.<sup>135–137</sup> Heat capacity data (Figure 9) revealed an extremely large low-temperature specific heat coefficient of  $\gamma = 8 \text{ J mol}^{-1} \text{ K}^{-2}$ ,<sup>135–137</sup> and the also determined Kondo temperature was measured to be  $T_K = 1$  K.<sup>134,136</sup> Further experiments showed that the antiferromagnetism is very sensitive to magnetic field, pressure, and strain.<sup>137–140</sup> In addition, below  $T_N$  very broad magnetic neutron-diffraction peaks were observed.<sup>141,142</sup>

YbCuAl crystallizes with a ZrNiAl type structure<sup>143–145</sup> (space group  $P\bar{6}2m$ ), was first discovered by Dwight et al. in 1968<sup>144</sup> and shows mixed-valent character for Yb in this compound.<sup>146,147</sup> Down to  $T = 30$  K, the susceptibility follows the Curie–Weiss law, while at the lowest temperatures it is marked by a saturation at a high level. The observed intermediate valence is close to 3, and YbCuAl can be described as a heavy Fermion compound with a high value of



**Figure 9.** Low-temperature specific heat  $C_p(T)$  of YbBiPt between 0.09 and 0.85 K. Inset: Same data as  $C_p/T$  vs  $T$ . Reprinted with permission from ref 136.

its specific heat coefficient ( $\gamma = 260 \text{ mJ mol}^{-1} \text{ K}^{-2}$ ).<sup>148,149</sup> For more details, we refer to the review of Bonville in 1988, who summarized the results of the different physical experiments on YbCuAl together with other ytterbium compounds.<sup>149</sup>

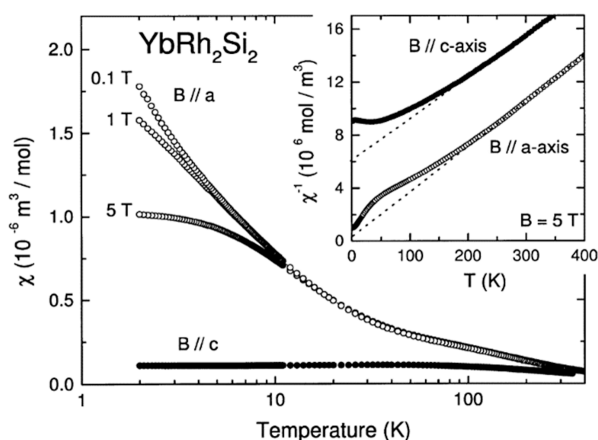
### Further Ternary Compounds

One of the most intensively studied ytterbium compounds is YbRh<sub>2</sub>Si<sub>2</sub>. Over the last 20 years, more than 350 publications have been published on this silicide. It was first described by Rossi et al. in 1979,<sup>22</sup> adopts the tetragonal ThCr<sub>2</sub>Si<sub>2</sub> type structure (space group  $I4/mmm$ ), and has a (theoretically) three-dimensional Fermi surface.<sup>150–154</sup> Antiferromagnetic ordering was detected at very low temperatures, with the Néel temperature being  $T_N = 70$  mK. Yet, the nature of this order is still not clear and was further examined with different techniques such as neutron scattering,<sup>155</sup> static susceptibility,<sup>156–157</sup> <sup>29</sup>Si NMR,<sup>157</sup> or ESR.<sup>158–160</sup> (vide infra). Ferromagnetic fluctuations cause incommensurate antiferromagnetic correlations, which were detected by these investigations above 70 mK.

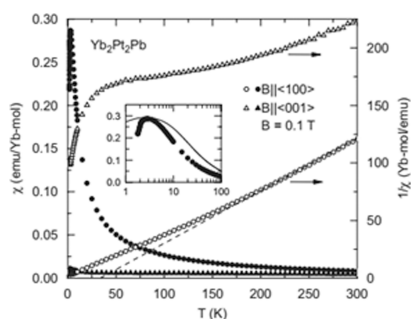
YbRh<sub>2</sub>Si<sub>2</sub> exhibits a non-Fermi-liquid behavior. The ordered Yb<sup>3+</sup> moment is very small, with only  $2 \times 10^{-3} \mu_B/\text{Yb}$  atom (Figure 10). The characteristic Kondo-temperature is  $T_K = 25$  K.<sup>161</sup> A quantum critical point is suggested for the suppression of  $T_N$  at ambient pressure by a magnetic field of only  $B_c = 60$  mT.<sup>162</sup> The substitution of Si by Ge shifts the quantum critical point to lower fields,<sup>156,163</sup> whereas Co substitution leads to ferromagnetism.<sup>164,165</sup> In addition, superconductivity was reported for YbRh<sub>2</sub>Si<sub>2</sub>, thus leading to the assumption of a strong coupling of electronic and nuclear magnetism.<sup>166,167</sup> Reasons for this superconductivity as well as the mentioned nature of ordering and the quantum critical point itself remain unclear and are the subject of further research.<sup>168</sup>

One of the remarkable trivalent ytterbium compounds is the Shastry–Sutherland<sup>170–172</sup> phase Yb<sub>2</sub>Pt<sub>2</sub>Pb, originally reported in 1999.<sup>47</sup> Yb<sub>2</sub>Pt<sub>2</sub>Pb adopts the Er<sub>2</sub>Au<sub>2</sub>Sn type structure (space group  $P4_2/mmm$ ). Later property studies were performed on single crystals which were grown from lead flux (Yb:Pt:Pb = 5:4:40 starting composition).<sup>173</sup> Orientation-dependent magnetic susceptibility measurements showed a high anisotropy with an about 30 times larger value for  $\chi_{(100)}$  as compared to  $\chi_{(001)}$  (Figure 11).<sup>174</sup> The Néel temperature of Yb<sub>2</sub>Pt<sub>2</sub>Pb is  $T_N = 2.07$  K.<sup>173</sup>





**Figure 10.** Temperature dependence of the magnetic and inverse magnetic susceptibility of  $\text{YbRh}_2\text{Si}_2$  measured parallel to the  $a$  and  $c$  axis with different applied fields. Reprinted with permission from ref 169.



**Figure 11.** Temperature dependence of the magnetic and inverse magnetic susceptibility of  $\text{Yb}_2\text{Pt}_2\text{Pb}$  measured parallel (100) and (001). Reprinted with permission from ref 174.

Besides,  $\text{Yb}_2\text{Pt}_2\text{Pb}$  was studied by specific heat and resistivity measurements and with respect to its crystal field parameters obtained by inelastic neutron scattering data. All these property studies were supported by quantum chemical calculations for a better understanding of the structure–property relationships. The complete property data of  $\text{Yb}_2\text{Pt}_2\text{Pb}$  has recently been reviewed,<sup>87</sup> and we refer to this broad overview for more details.

Magnetization data support the anisotropic behavior observed in the orientation-dependent susceptibility measurements.  $\text{Yb}_2\text{Pt}_2\text{Pb}$  shows multiple metamagnetic steps in both the (100) and (110) directions.<sup>175</sup> The maximum magnetization observed in (100) at 5 T is around  $2.4 \mu_B$  per ytterbium atom, significantly reduced when compared with the theoretical value of  $4 \mu_B$  ( $g \times J$ ). The magnetic structure of  $\text{Yb}_2\text{Pt}_2\text{Pb}$  was studied from powder and single crystal neutron diffraction data.<sup>176</sup> The ordered ytterbium magnetic moments were described in a  $5 \times 5 \times 1$  magnetic supercell.

$\text{YbAl}_3\text{C}_3$  is another example of isotropic exchange interaction and crystallizes with the hexagonal  $\text{ScAl}_3\text{C}_3$  type structure (space group  $P6_3/mmc$ ). The ytterbium atoms form a triangular lattice, then a structural phase transition to an orthorhombic phase (own structure type) occurs at  $T_S = 80$  K.<sup>177–179</sup> Above this temperature, Curie–Weiss behavior with an effective magnetic moment of  $\mu_{\text{eff}} = 4.66 \mu_B$  was observed, underlining the trivalent state of the ytterbium atoms. The calculated Weiss constant of  $\Theta_p = -120$  K indicates

antiferromagnetic interactions in the paramagnetic regime.<sup>177,178</sup> Below  $T_S$ , no long-range magnetic order was visible by measurements of the magnetization or specific heat. It has been proposed that the anisotropic Kramers doublet of  $\text{YbAl}_3\text{C}_3$  forms an antiferromagnetic dimer ground state with a single-triplet energy gap of about 15 K in a relatively narrow field and the 4f electrons are expected to behave as a  $S = 1/2$  spin system at low temperatures, thus suggesting only weaker interdimer interaction in the ytterbium triangles.<sup>177,178,180–182</sup>

$\text{YbCo}_2\text{Zn}_{20}$  is a member of the  $\text{CeCr}_2\text{Al}_{20}$  type family (space group  $Fd\bar{3}m$ ). The high temperature susceptibility follows the Curie–Weiss law with an effective magnetic moment near the value of a free  $\text{Yb}^{3+}$  ion ( $\mu_{\text{eff}} = 4.54 \mu_B$ ). No magnetic ordering down to 20 mK was observed, while the recorded magnetization isotherms at low temperatures indicate a metamagnetic transition. At lowest temperatures, the magnetic ground state is enhanced by the Pauli susceptibility. By heat capacity measurements, a large Sommerfeld coefficient of  $\gamma = 7.9 \text{ J}^{-1} \text{ mol}^{-1} \text{ K}^{-2}$  was observed. Kondo interactions were confirmed via the  $T^2$  resistivity, resulting from the  $A$  coefficient ( $A = 165 \mu\Omega \text{ K}^{-2}$ ). This classifies  $\text{YbCo}_2\text{Zn}_{20}$  as a superheavy electron metal. For further detail, we refer to the literature and reviews.<sup>88,183–186</sup>

Stoichiometric  $\text{YbInCu}_4$  adopts the  $\text{MgCuSn}_4$  type structure<sup>187,188</sup> (cubic  $C15B$ , space group  $F\bar{4}3m$ );<sup>189</sup> however, the disordered structure (solid solution)  $\text{Yb}_{1-x}\text{In}_x\text{Cu}_2$  ( $C15$ ,  $\text{MgCu}_2$  type, space group  $Fd\bar{3}m$ ) has also been observed.<sup>190,191</sup> It is one of the few Yb compounds ( $\text{YbPd}_2\text{Al}_3$  is another one<sup>67</sup>) that exhibits a first-order isostructural valence transition (VT) with the transition temperature  $T_{\text{trans}} = 42$  K.<sup>189–191</sup> In the high-temperature regime, the Yb valence has been reported to be +2.9 and suddenly drops to +2.74 when cooling below  $T_{\text{trans}}$ .<sup>192</sup> At the same time, the magnetic ground state undergoes a transition from a local moment susceptibility of  $4.5 \mu_B$  to temperature-independent Pauli paramagnetism below  $T_{\text{trans}}$ .<sup>193</sup> No change in the structure is observed during the transition, although the unit cell volume increases by  $\sim 0.5\%$ .<sup>194,195</sup>

Furthermore, the transition was confirmed by synchrotron radiation X-ray absorption, X-ray emission spectroscopy, inelastic neutron scattering (vide infra),<sup>196–200</sup> and abrupt changes in other physical properties like electrical resistivity, specific heat capacity, and nuclear spin-relaxation rate (vide infra).<sup>198,201,202</sup> The transition is also accompanied by a drastic change of the Kondo temperature ( $T_K$ ) from 20 to  $\sim 400$  K in the low-temperature phase.<sup>199,203</sup>

In Yb-based heavy Fermion systems,  $\beta\text{-YbAlB}_4$  was the first discovered superconductor with the transition temperature  $T_C = 80$  mK.<sup>204,205</sup> The results of transport and thermodynamic properties suggest non-Fermi liquid behavior in the normal  $\beta\text{-YbAlB}_4$  state.<sup>203,206</sup> It adopts the orthorhombic  $\text{ThMoB}_4$  type structure (space group  $Cmmm$ ) with the Yb and Al atoms being in the  $ab$  plane between two B layers of hepta- and hexagonal boron rings (vide ultra).<sup>93</sup> This structural environment and the strong observed valence fluctuations<sup>207</sup> are essential for understanding the properties of  $\beta\text{-YbAlB}_4$ . Within the  $ab$  plane, the Kondo hybridization was discussed to be isotropic, in contrast to the anisotropy between the crystal  $c$  axis and the  $ab$  plane.<sup>208</sup> This may explain the observed  $T/B$  scaling of the magnetic susceptibility and the intrinsic quantum criticality.<sup>209–214</sup>

Another example of an intermediate valence ytterbium compound is  $\text{YbNiAl}_4$ , which adopts the  $\text{YNiAl}_4$  type structure

(space group  $Cmcm$ ). The temperature dependence of the magnetic susceptibility is compatible with the ICF model. At  $T = 4.2$  K the determined valence of the Yb atoms is +2.1, in contrast to +3 at  $T = 1000$  K. Resistivity measurements revealed a Fermi-liquid behavior, due to the observed  $AT^2$  variation below 60 K. At room temperature, it shows a metallic-like behavior. The heat capacity measurements gave no hints for any magnetic ordering down to 1.9 K. For the electronic specific heat capacity, a value of  $\gamma = 16$  mJ mol<sup>-1</sup> K<sup>-2</sup> was determined, so that a heavy Fermion state can be excluded.<sup>23,215</sup> In addition, various transport measurements (thermal and electrical) were analyzed.<sup>216</sup> In-situ high-pressure studies show an increase of the ytterbium valence with increasing pressure for YbNiGa<sub>4</sub> and YbNiIn<sub>4</sub>.<sup>217</sup>

Also, the complex structure of Yb<sub>4</sub>Ga<sub>24</sub>Pt<sub>9</sub>, a platinumide, shows intermediate ytterbium valence. Hard X-ray photoelectron spectra and magnetic susceptibility measurements point to an ytterbium valence of  $\sim 2.5$ .<sup>218</sup> The stannide YbRhSn<sub>2</sub>,<sup>219</sup> and the solid solution YbIn<sub>1-x</sub>Au<sub>1+x</sub> ( $x = 0-0.3$ )<sup>220</sup> show comparable behavior.

An intermediate ytterbium valence can also be induced by isovalent substitution. The phosphide Yb<sub>2</sub>Ni<sub>12</sub>P<sub>7</sub> exhibits a heavy-Fermion paramagnetic ground state and switches to intermediate valence in Yb<sub>2</sub>Ni<sub>12</sub>As<sub>7</sub>.<sup>221</sup>

## INELASTIC NEUTRON SCATTERING

Inelastic neutron scattering is often used to probe crystal field excitations which allow drawbacks on the crystal electric field (CEF) splitting for a certain atom and magnetic phase transitions. In contrast to elastic scattering, in this case, the energy difference of the scattered neutrons is detected. Therefore, it is a useful tool to obtain information about the ground state and the magnetic exchange interactions of, in this case, the Yb atoms.<sup>234</sup>

### Binary Compounds

One of the first compounds that has been studied by inelastic neutron scattering was YbBe<sub>13</sub> (NaZn<sub>13</sub> type, space group  $Fm\bar{3}c$ ).<sup>235</sup> The spectra were recorded on a polycrystalline sample between 1.2 and 300 K. Since YbBe<sub>13</sub> undergoes antiferromagnetic ordering at  $T_N = 1.28$  K,<sup>236</sup> the single line observed at 1.2 K was interpreted as a magnon. The spin fluctuations associated with the magnetic ordering, however, persist for quite a large temperature range and are still observable at 10 K. At 39 K, the CEF splitting can be observed with  $\Gamma_7$  being the ground state and  $\Gamma_8$  (3.20 meV) and  $\Gamma_6$  (4.39 meV) being the next excited states. These observations contradict the results from the literature where a different sequence was reported; however, all agree on the  $\Gamma_7$  ground state.<sup>236,237</sup>

For YbPd (CsCl type, space group  $Pm\bar{3}m$ ) and Yb<sub>3</sub>Pd<sub>4</sub> (Pu<sub>3</sub>Pd<sub>4</sub> type, space group  $R\bar{3}$ ), inelastic neutron scattering has been used to probe the proposed magnetic transitions in these intermediate valent compounds and to determine the CEF splitting to gain information about the ground state. For YbPd,  $\Gamma_8$  was determined to be the ground state with  $\Gamma_7$  being 4.75 meV and  $\Gamma_6$  being 12.3 meV higher in energy. This scheme is in line with trivalent Yb in a cubic environment. At 1.3 K, an additional contribution with magnetic origin is observed for YbPd. For Yb<sub>3</sub>Pd<sub>4</sub>, the first excitation was shifted to lower energy (4.08 meV) and is more distinct compared to the one in YbPd. It is interesting to note that despite Yb<sub>3</sub>Pd<sub>4</sub> being trigonal (26 independent CF parameters are expected—note,

the referenced paper states that Yb<sub>3</sub>Pd<sub>4</sub> is triclinic; this statement has been corrected) the observed transitions still can be interpreted by a cubic site symmetry. Finally, measurements between 1.3 and 6 K revealed a strong inelastic line at low temperatures that softens toward the proposed magnetic phase transition near 3 K. An abrupt change of the line width at  $T = 3.1$  K indicates that this excitation can be attributed to a magnon in the magnetically ordered state.<sup>238</sup>

### Ternary Compounds

To understand the valence fluctuating behavior of YbInAu<sub>2</sub> (MnCu<sub>2</sub>Al type, space group  $Fm\bar{3}m$ ), neutron scattering experiments were conducted and compared to the results of YbAl<sub>3</sub> (Cu<sub>3</sub>Au type, space group  $Pm\bar{3}m$ ) which exhibits similar physical properties. The data obtained at temperatures near the susceptibility maxima ( $\sim 80$  K for YbInAu<sub>2</sub> and  $\sim 120$  K for YbAl<sub>3</sub>) could be fitted with a single spectral function. At 5 K, however, the data of YbAl<sub>3</sub> needs more components while the data of YbInAu<sub>2</sub> still can be fitted with a single function. The differences are also visible in the phonon scattering (high angle data). Here, a well-defined phonon mode is visible for YbAl<sub>3</sub> in contrast to YbInAu<sub>2</sub>.<sup>239</sup>

As mentioned before, Yb<sub>1-x</sub>In<sub>x</sub>Cu<sub>2</sub> (MgCu<sub>2</sub> type, space group  $Fd\bar{3}m$ ) shows a temperature-dependent valence phase transition. It changes from a nearly trivalent state at high temperatures to a mixed-valent Pauli paramagnetic state at low temperatures. A sample with  $x = 0.5$  was investigated at  $T = 30$  and 60 K, above and below the transition temperature of  $T = 40$  K. The spin dynamics in the mixed-valent phase ( $T = 30$  K) can be modeled with a Lorentzian power spectrum with the center at  $E = 40$  meV.<sup>240</sup>

Also, different YbT<sub>2</sub>X<sub>2</sub> compounds were investigated by inelastic neutron scattering. Among the first investigations was the work on YbT<sub>2</sub>Si<sub>2</sub> for  $T = \text{Fe, Co, and Ni}$  (ThCr<sub>2</sub>Si<sub>2</sub> type, space group  $I4/mmm$ ).<sup>241</sup> The performed experiments showed that YbCo<sub>2</sub>Si<sub>2</sub> exhibits three excited states at  $\sim 4$ , 12.5, and 30.5 meV above the ground state, while for YbNi<sub>2</sub>Si<sub>2</sub>, two clear transitions at  $\sim 7.5$  and 13.5 meV are visible alongside a shoulder at lower energy of the 7.5 meV peak. For YbFe<sub>2</sub>Si<sub>2</sub>, no clear inelastic signals are observed.<sup>241</sup> Taking the experimentally determined energies for YbCo<sub>2</sub>Si<sub>2</sub> into account, the CF level scheme could be derived, indicating  $\Gamma_7$  to be the ground state.<sup>242</sup> YbMn<sub>2</sub>Si<sub>2</sub> (ThCr<sub>2</sub>Si<sub>2</sub> type, space group  $I4/mmm$ ) has the peculiarity that it shows two antiferromagnetic transitions at  $T_{N,1} = 526$  and  $T_{N,2} = 30$  K originating from the Mn atoms.<sup>243</sup> Determination of the magnetic structure showed that, below  $T_{N,1}$ , the Mn atoms show a + - + - sequence, while below  $T_{N,2}$ , a + - - + magnetic substructure arises, which leads to a doubling of  $c$  in the magnetic unit cell. At 1.5 K also the Yb atoms are antiferromagnetically ordered, however, with their spins orthogonal to the Mn spins.<sup>244</sup> Above 30 K, only one Yb site is present which exhibits a doublet ground state as well as three excited states at 4.65, 11.89, and 15.05 meV. Below 30 K, however, a second Yb site emerges due to the antiferromagnetically ordered Yb atoms. This leads to two additional signals in the inelastic neutron spectrum of YbMn<sub>2</sub>Si<sub>2</sub> at 2.5 K. These transitions can be associated with a different CEF scheme for the Yb2 site with energy levels at 7.24, 8.76, and 16.4 meV.<sup>245,246</sup> For the heavy Fermion compound YbRh<sub>2</sub>Si<sub>2</sub> (ThCr<sub>2</sub>Si<sub>2</sub> type, space group  $I4/mmm$ ), inelastic neutron scattering experiments conducted at 1.5 K are also dominated by the CEF excitations of the Yb<sup>3+</sup> ion. Besides a ground state doublet, three excited state

doublets were observed at higher energies of 17, 25, and 43 meV, in line with the CF theory for a tetragonal system.<sup>247</sup> Subsequent calculations showed that the ground state is a  $\Gamma_6$  Kramers doublet.<sup>248</sup> For  $\text{YbIr}_2\text{Si}_2$ , both modifications ( $\text{CaBe}_2\text{Ge}_2$  type,  $P4/nmm$ , and  $\text{ThCr}_2\text{Si}_2$  type,  $I4/mmm$ ) were investigated by inelastic neutron scattering experiments at 1.5 K. As for  $\text{YbRh}_2\text{Si}_2$ , a doublet ground state and three excited doublet states are expected. For the body-centered polymorph, energies of 0, 18, 25, and 36 meV were observed, while for the primitive polymorph they are slightly lower in energy at 0, 15, 22, and 33 meV.<sup>249</sup> These values are in good agreement with the calculated values for both  $\text{YbRh}_2\text{Si}_2$  and  $\text{YbIr}_2\text{Si}_2$ .<sup>250</sup>

## XANES/XAS

X-ray absorption spectroscopy (XAS) describes the overall measurement of the absorption spectrum of electrons, which interact with polychromatic X-ray photons and absorb characteristic wavelengths. The XA spectrum can be separated into two parts. First, the area at higher photon energy than the absorption, called extended X-ray absorption fine structure (EXAFS) which contains information about the local environment of the absorbing atom. The second area is at the absorption energy, the X-ray absorption near edge structure (XANES) or near edge X-ray absorption fine structure (NEXAFS), which gives insight into the electronic properties, e.g., the oxidation state of the absorbing atom. In the following chapter, examples of classical intermetallic Yb-compounds as well as Zintl phases will be addressed with regards to their respective valence in the XAS and XANES measurements.

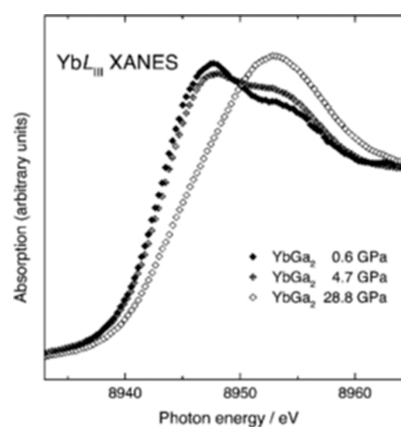
### Binary Compounds

The binary compounds  $\text{YbAl}_2$  ( $\text{MgCu}_2$  type, space group  $Fd\bar{3}m$ ) and  $\text{YbAl}_3$  ( $\text{Cu}_3\text{Au}$  type, space group  $Pm\bar{3}m$ ) are among the most investigated intermetallics. Both have one crystallographic Yb position in the crystal structure. One of the earliest X-ray absorption measurements of  $\text{YbAl}_2$  was done by Beaurepaire, Kappler, and Krill in 1986.<sup>251</sup> In their work,  $\text{YbAl}_2$  was measured at room temperature under pressures of 10 and 30 kbar. At room temperature,  $\text{YbAl}_2$  has a valence of +2.3, whereas under pressure the valence rises to +2.6 at 30 kbar (pressure-induced oxidation). Due to the width of the 2p core hole of 4.6 eV, the  $\text{Yb}^{2+}$  and  $\text{Yb}^{3+}$   $L_{\text{III}}$  edges could not be resolved, but the two contributions are separated by about 6 eV.<sup>251</sup> Dallera et al. measured the pressure dependence of the Yb  $L_{\text{III}}$  edge in  $\text{YbAl}_2$  up to 385 kbar using the partial fluorescence yield mode of the X-ray absorption (PFY-XAS), giving a higher resolution of the spectra compared to the conventional measurement mode. The absorption spectra show the valence shift from a valence in  $\text{YbAl}_2$  of +2.25 at ambient pressure to a valence of +2.9 at 385 kbar. At pressures of 170 and 385 kbar, the spectra of  $\text{YbAl}_2$  become similar to  $\text{Yb}_{1-x}\text{In}_x\text{Cu}_2$ , which has a stated valence of +2.96, indicating a valence close to +3.<sup>252</sup>

XAS spectra of  $\text{YbAl}_3$  were measured at 10 and 300 K, showing the Yb 3d  $\rightarrow$  4f transition and indicating an open 4f shell in the ground state. A decrease in the intensity of 11% at 10 K compared to 300 K is additionally reported by Tjeng et al., suggesting a decrease in the valence of the Yb atom, which is also shown in the photoemission spectroscopy.<sup>253</sup> The pressure-induced valence change of  $\text{YbAl}_3$  was measured by PFY-XAS starting at ambient pressure and the valence of +2.75 for  $\text{YbAl}_3$ , which subsequently rises alongside the increasing

pressure. At 9 GPa, the valence is +2.82, then +2.86 at 20 GPa, and finally +2.93 at an applied pressure of 38 GPa. The incident energy for  $\text{Yb}^{2+}$  is 8.942 keV,  $\text{Yb}^{3+}$  resonates at a slightly higher energy of 8.950 keV.<sup>254</sup> By changing from bulk materials to nanoparticles, the valence of  $\text{YbAl}_3$  changes from +2.86 to +2.70 after mechanical milling in a planetary high-energy ball milling system for 120 h. The reference  $\text{Yb}_2\text{O}_3$  indicates the  $\text{Yb}^{3+}$  peak at 8946.5 eV, and the spectra of the milled  $\text{YbAl}_3$  shows an increase of the  $\text{Yb}^{2+}$  contribution at 8941 eV. The valences for 20 and 70 h milling time are +2.74 and +2.70, respectively. A cause of the valence change is the size effect stated. The milled  $\text{YbAl}_3$  particles possess two types of Yb atoms, the trivalent in the bulk of the sample and the ones near the surface having a variation of their coordination number, leading to divalent Yb. Another reason is the disorder introduced in the ball milling process, which is particularly visible in the X-ray diffraction data.<sup>255</sup>

$\text{YbGa}_2$  crystallizes at ambient pressure in the hexagonal  $\text{CaIn}_2$  type structure ( $P6_3/mmc$ ), at higher pressures above 22 GPa a phase transition into the  $\text{UHg}_2$  type ( $P6/mmm$ ) occurs. XANES measurements of the Yb  $L_{\text{III}}$ -edge show a valence increase from  $\text{Yb}^{2+}$  at ambient pressure to  $\text{Yb}^{3+}$  in the high-pressure structure. At low pressures, two peaks are visible, wherein the peak at higher energy represents the  $\text{Yb}^{3+}$  and is within 1 eV to the reference  $\text{Yb}_2\text{O}_3$ , and the second peak is 7 eV lower, corresponding to  $\text{Yb}^{2+}$ . Measurements were taken at 0.6, 4.7, and 28.8 GPa, showing a continuous increase in the intensity of  $\text{Yb}^{3+}$  and decrease of the  $\text{Yb}^{2+}$  white line until the phase transition occurs and only  $\text{Yb}^{3+}$  is present (Figure 12). In



**Figure 12.** Yb  $L_{\text{III}}$  XANES spectra of  $\text{YbGa}_2$  at different pressures. The white lines at low energies are attributed to  $\text{Yb}^{2+}$ , and the lines at higher energies are attributed to  $\text{Yb}^{3+}$ . Figure reproduced with permission from ref 256. Copyright 2001 John Wiley & Sons, Inc.

the high-pressure modification,  $\text{YbGa}_2$  possesses a two-dimensional polyanion, compared to the Ga network found in the normal pressure phase. The two-dimensional arrangement can then be supported with the additional electrons from the Yb atoms.<sup>256</sup>

The hexagonal phase  $\text{Yb}_5\text{Si}_3$  crystallizes in the  $\text{Mn}_5\text{Si}_3$  type structure ( $P6_3/mcm$ ) and possesses two distinct crystallographic Yb positions (4d and 6g). Alongside XANES, Mössbauer spectroscopic investigations and magnetic measurements were also conducted. The magnetic measurements indicated that only trivalent Yb is present, in contrast to the Mössbauer spectrum proposing 60% of  $\text{Yb}^{3+}$ . XANES measurements were performed at 10 and 300 K; these show

two white lines at 8943 and 8935 eV corresponding to  $\text{Yb}^{3+}$  and  $\text{Yb}^{2+}$ , respectively. The calculated valence from the spectra is +2.88, which is higher than the expected value of +2.6 if the Yb atoms on the 6d site were purely trivalent and the ones on the 4g site divalent. The authors suggested that a hybridization of the 4f electrons is occurring, whose strength is different for the two different crystallographic sites. The Yb atoms on the 4d site are more strongly influenced than the Yb atoms on 6g.<sup>257</sup>

$\text{Yb}_5\text{Ge}_4$  ( $\text{Sm}_5\text{Ge}_4$  type, space group  $Pnma$ ) is reported to be a mixed valence compound with a valence of +2.42, showing the white band at 8948 eV and a big shoulder at 8941 eV, indicating  $\text{Yb}^{3+}$  and  $\text{Yb}^{2+}$ , respectively.<sup>258</sup>

### Equiatomic Ternary Compounds

The heavy Fermion compound  $\text{YbCuAl}$  ( $\text{ZrNiAl}$  type, space group  $P\bar{6}2m$ ) is described with a Yb valence of nearly +3. With increasing pressure, the valence of Yb can be increased, and with lowering the temperature, the Yb becomes more divalent as shown by XES measurements.<sup>225</sup>  $\text{YbAuIn}$  also crystallizes in the hexagonal  $\text{ZrNiAl}$  type structure with one crystallographic Yb position. X-ray absorption measurements of  $\text{YbAuIn}$  were taken at 15–18 and 300 K at ambient pressure, resulting in an intensive white line at 8941 eV corresponding to a divalent Yb, but also a weak shoulder at 8949 eV showing a small amount of trivalent Yb. By fitting the spectra and excluding  $\text{Yb}_2\text{O}_3$  impurities, a valence of +2.17 was determined.<sup>259</sup>  $\text{YbGaGe}$  possesses two crystallographically different Yb positions and crystallizes in the hexagonal  $\text{YPtAs}$  type structure ( $P6_3/mmc$ ). The compound shows a white band at around 8940 eV and a small shoulder around 8950 eV, indicating a divalent system with oxidic impurities, which is in line with the magnetic measurements.<sup>260</sup>

### Further Ternary Compounds

$\text{YbRh}_2\text{Si}_2$  ( $\text{ThCr}_2\text{Si}_2$  type, space group  $I4/mmm$ ) has a temperature- and magnetic-field-dependent Yb valence, which is decreasing with lower temperature and increasing alongside higher magnetic fields. To measure the influence of the two parameters, XAS measurements were conducted by Nakai et al. The Yb  $L_{\text{III}}$  absorption spectra show a shoulder at 8940 eV appearing at decreasing temperature and an intensive white line at 8946 eV, indicating the trivalent Yb, whose intensity decreases with lower temperatures. Above 200 K, the valence was determined to be +2.96, while below 25 K, a valence of +2.92 was reported. In contrast, the valence change of Yb was measured by increasing magnetic field at 2 K. At 0 T, the valence is +2.92, and it rises to +2.93 monotonically with stronger magnetic fields up to 33 T.<sup>261</sup>

X-ray absorption spectra of  $\text{YbCu}_2\text{Si}_2$ ,  $\text{YbCu}_2\text{Ge}_2$  ( $\text{ThCr}_2\text{Si}_2$  type, space group  $I4/mmm$ ), and  $\text{YbPt}_2$  ( $\text{MgCu}_2$  type, space group  $Fd\bar{3}m$ ) were measured at 300 K, showing a pure trivalent Yb in  $\text{YbPt}_2$ , divalent Yb in  $\text{YbCu}_2\text{Ge}_2$ , and a mixed valence occurring in  $\text{YbCu}_2\text{Si}_2$  with a higher amount of  $\text{Yb}^{3+}$  than  $\text{Yb}^{2+}$ . It was also stated that the Yb valence in  $\text{YbCu}_2\text{Si}_2$  does not show any signs of a temperature dependency.<sup>262</sup> Similar to  $\text{YbCu}_2\text{Ge}_2$ ,  $\text{YbAl}_2\text{Ga}_2$  possesses a white line at 8936 eV and a resulting valence of  $\text{Yb}^{2+}$ .<sup>263</sup>  $\text{YbMn}_2\text{Ge}_2$  ( $\text{ThCr}_2\text{Si}_2$  type, space group  $I4/mmm$ ) is also described as a mixed valent compound with a Yb valence of +2.40 at 300 K. This compound shows not only a negative thermal expansion, but also a temperature dependence of the Yb valence. With the increase of the temperature, the white line at 8943 eV decreases while the one at 8949 eV increases, corresponding to the  $\text{Yb}^{2+}$  and  $\text{Yb}^{3+}$

respectively, leading to a valence of +2.82 at 700 K. Also, the valence of Yb can be influenced by applying pressure; e.g., at a pressure of 1.4 GPa, the Yb valence is +2.82.<sup>264</sup>

The  $\text{YbT}_2\text{X}_{20}$  compounds with  $T = \text{Cr, Ti, V, Co, Fe, Ir, Rh}$  and  $X = \text{Al, Zn}$  all crystallize in the cubic  $\text{CeCr}_2\text{Al}_{20}$  type structure (space group  $Fd\bar{3}m$ ). These structures are heavy Fermion compounds and can possess a particularly high specific heat coefficient. XAS measurements were taken of  $\text{YbRh}_2\text{Zn}_{20}$  and  $\text{YbIr}_2\text{Al}_{20}$  at 2, 250, and 290 K, respectively, with additional measurements at 2 K with an applied external field of 10 T. The absorption spectra of  $\text{YbRh}_2\text{Zn}_{20}$  have two peaks, the white line at 8950 eV indicating  $\text{Yb}^{3+}$  and a shoulder at 8940 eV originating from  $\text{Yb}^{2+}$ . The valence at 250 K is stated as +2.89 and is decreasing with decreasing temperature, especially below 100 K, reaching a valence of +2.87 at 20 K. The same trend is visible for  $\text{YbIr}_2\text{Al}_{20}$ , also having a decreasing Yb valence with decreasing temperatures below 90 K; above 100 K, the valence is nearly constant. As for the measurements under an applied field, only changes in the valence are present below 80 K.<sup>265</sup> In contrast,  $\text{YbCo}_2\text{Zn}_{20}$  was investigated by XAS measurements, showing no indication of temperature or magnetic field dependency.<sup>266</sup>  $\text{YbFe}_2\text{Zn}_{20}$  and Cd-substituted  $\text{YbFe}_2\text{Zn}_{20-x}\text{Cd}_x$  with  $x = 0-1.4$  were analyzed by Fahl et al. Their XANES studies of the Yb  $L_{\text{III}}$  absorption show only the white line at 8946 eV corresponding to  $\text{Yb}^{3+}$  and no additional shoulder, making all compounds formally trivalent.<sup>267</sup>

For the valence transition compounds  $\text{Yb}_{1-x}\text{In}_x\text{Cu}_2$  ( $x = 0.5$ ), the valence transition temperature is stated to be  $T_{\text{trans}} = 42$  K where the valence changes from +2.98 to +2.85, as calculated from the XANES measurements at 300 and 7 K, respectively.<sup>268</sup> Another study on  $\text{Yb}_{0.5}\text{In}_{0.5}\text{Cu}_2$  by Yamaoka et al. focused on the pressure-induced valence transition. In their work, X-ray absorption spectra of  $\text{Yb}_{0.5}\text{In}_{0.5}\text{Cu}_2$  were taken at 300 and 13 K under pressure. At 300 K,  $\text{Yb}_{0.5}\text{In}_{0.5}\text{Cu}_2$  shows no pressure dependent valence; at 13 K on the other hand, it presents an increase of valence at pressures of 3.1 GPa and above.<sup>269</sup> The XAS spectra of  $\text{YbAgCu}_4$  ( $\text{MgCu}_4\text{Sn}$  type, space group  $F\bar{4}3m$ ) measured at 10 and 300 K by Lawrence et al. show the white line at 8947 eV and a shoulder at 8940 eV, corresponding to  $\text{Yb}^{3+}$  and  $\text{Yb}^{2+}$ , respectively. Also, they are stating that the valence of  $\text{YbAgCu}_4$  is nearly +3 and changes continuously with temperature.<sup>270</sup> Sarrao et al. investigated the Yb valence in the  $\text{YbTCu}_4$  series ( $\text{MgCu}_4\text{Sn}$  type, space group  $F\bar{4}3m$ ) with  $T = \text{Mg, Ag, Au, Zn, Cd, Tl}$  by XAS of the Yb  $L_{\text{III}}$  line between 20 and 300 K, showing the Mg compound with the lowest Yb valence of +2.64 at 20 K and +2.69 at 300 K, followed by  $\text{YbTlCu}_4$  with +2.75 to +2.82,  $\text{YbCdCu}_4$  +2.76 to +2.82,  $\text{YbZnCu}_4$  +2.84 to +2.88,  $\text{YbAgCu}_4$  +2.86 to +2.93, and finally  $\text{YbAuCu}_4$  with a valence of +2.95 at 20 K and +2.96 at 300 K.<sup>240</sup> XANES studies of  $\text{YbPtIn}_4$  and  $\text{YbPdIn}_4$  ( $\text{YNiAl}_4$  type, space group  $Cmcm$ ) were reported, stating both compounds containing mixed valent Yb.<sup>271</sup>

### Zintl Phases

The Zintl compounds  $\text{Yb}_{14}\text{TSb}_{11}$  ( $T = \text{Zn, Mn, Mg; Ca}_{14}\text{AlSb}_{11}$  type, space group  $I4_1/acd$ ) exhibit besides intriguing magnetic also highly important thermoelectric properties (see *thermoelectric properties*). Using the Zintl counting rules, the Mn-containing compounds only have  $\text{Yb}^{2+}$  cations, while the Zn and Mg compound should also contain one  $\text{Yb}^{3+}$ . To verify the valence of the Yb atoms in these compounds, XANES measurements were taken by He et al. The absorption spectra of all three compounds indicate the

presence of both  $\text{Yb}^{2+}$  and  $\text{Yb}^{3+}$ , but the overall valence is close to  $\text{Yb}^{2+}$ . Additionally, it is visible that the Mg and Mn compounds have a constant amount of  $\text{Yb}^{3+}$ , but in the Zn compound, the valence is highly temperature-dependent. The formation of  $\text{Yb}^{3+}$  is explained by introducing disorder in the structure of  $\text{Yb}_{14}\text{TSb}_{11}$ .<sup>272</sup>

### ■ X-RAY PHOTOELECTRON SPECTROSCOPY (XPS)

X-ray photoelectron spectroscopy (XPS) describes the analysis of a (solid) material, using the photoelectric effect, meaning the emission of an electron due to absorption of X-ray radiation. Typically, Al  $K\alpha$  is used as an X-ray source with a photon energy around 1486 eV. Due to the nature of the interaction between condensed matter and electrons, especially the short inelastic mean free path of, e.g., 1–2 nm for metallic systems, makes this analysis a rather surface-sensitive method.

#### Binary Compounds

$\text{YbAl}_2$  (MgCu<sub>2</sub> type, space group  $Fd\bar{3}m$ ) was measured by XPS in 1985 by Oh et al. at 120, 460, and 750 K. The spectra at the two lower temperatures show the peaks of both  $\text{Yb}^{2+}$  and  $\text{Yb}^{3+}$ , with the  $4f^{14} \rightarrow 4f^{13}$  and  $4f^{13} \rightarrow 4f^{12}$  bands, respectively. At higher temperatures, the intensity of the peak originating from the divalent Yb atoms is reduced, while the trivalent one increases, leading to valences of +2.32 at 120 K, +2.40 at 460 K, and finally +2.53 at 750 K.<sup>273</sup> In 2012, the valence of Yb in  $\text{YbAl}_2$  was analyzed by HAXPES using hard X-ray radiation of 7.94 keV for excitation at 20 K, allowing the Yb 3d core levels to be shown. In the spectrum recorded at 20 K, the Yb 3d core levels can be seen at 1515–1540 eV ( $3d_{5/2}$ ) and 1560–1585 eV ( $3d_{3/2}$ ). The spectra at 20 and 300 K both show  $\text{Yb}^{2+}$  and  $\text{Yb}^{3+}$  species, leading to a mean valence of +2.20, staying nearly unchanged below 300 K.<sup>116</sup>

The valence of  $\text{YbAl}_3$  (Cu<sub>3</sub>Au type, space group  $Pm\bar{3}m$ ) was analyzed using XPS by Buschow et al. in 1977. The Yb 4f spectra show a peak multiplet extending from 5 to 10 eV, corresponding to  $\text{Yb}^{3+}$ . Also, a spin doublet is visible from 0 to 3 eV, indicating divalent Yb, but no mean valence of the compound could be calculated from the measured spectra.<sup>274</sup> Later, in 1996,  $\text{YbAl}_3$  was measured by XPS, with a well-resolved doublet of divalent Yb and the multiplet of trivalent Yb. The mean valence of  $\text{YbAl}_3$  was determined to be +2.63 based on the recorded XPS spectra.<sup>275</sup> The nearly same results were reported by Joyce et al. in 2001, measuring  $\text{YbAl}_3$  around 20 K showing both Yb valence species in the spectra with a mean valence around +2.6.<sup>276</sup>

$\text{YbCu}_2$  and  $\text{YbAg}_2$  (both KHg<sub>2</sub> type, space group  $Imma$ ),  $\text{YbAu}_2$  (MoSi<sub>2</sub> type, space group  $I4/mmm$ ), and  $\text{YbNi}_5$  (CaCu<sub>5</sub> type, space group  $P6/mmm$ ) were measured by XPS to determine the valence states of ytterbium in these compounds. The measured spectra show the 4f and the 5p levels. As expected, the trivalent Yb compound  $\text{YbNi}_5$  has peaks only in the region of  $\text{Yb}^{3+}$  while  $\text{YbCu}_2$ , which is purely divalent, shows the two peaks corresponding to  $\text{Yb}^{2+}$ .  $\text{YbAg}_2$  on the other hand, exhibits a small trivalent signal due to oxidic impurities. According to the unit cell volume,  $\text{YbAu}_2$  should possess a trivalent Yb state, but the XPS measurements showed not only the  $\text{Yb}^{3+}$  peaks, but also the more intensive peaks of divalent Yb in the spectra, indicating  $\text{YbAu}_2$  to rather be an intermediate valence system, compared to  $\text{YbCu}_2$ .<sup>277</sup>

The valence fluctuating compound  $\text{YbB}_{12}$  (UB<sub>12</sub> type, space group  $Fm\bar{3}m$ ) was investigated at room temperature, showing the  $\text{Yb}^{2+}$  doublet at 0–3 eV and the  $\text{Yb}^{3+}$  multiplet between 5

and 12 eV. The mean valence of Yb was determined to be +2.9.<sup>278</sup>

Finally,  $\text{Yb}_4\text{As}_3$ ,  $\text{Yb}_4\text{Sb}_3$ , and  $\text{Yb}_4\text{SbBi}_3$  (all Th<sub>3</sub>P<sub>4</sub> type, space group  $I43d$ ) are also known to be valence fluctuating compounds. The XPS spectra, measured at room temperature, all show the peaks for divalent and trivalent Yb, although the  $\text{Yb}^{3+}$  peak in  $\text{Yb}_4\text{Bi}_3$  is of low intensity. After analyzing the Yb 4f spectra, the mean Yb valences were calculated to be +2.32, +2.34, and +2.10 for  $\text{Yb}_4\text{As}_3$ ,  $\text{Yb}_4\text{Sb}_3$ , and  $\text{Yb}_4\text{Bi}_3$ , respectively.<sup>279</sup>

#### Equiatomic Compounds

$\text{YbAuMg}$  crystallizes in the orthorhombic TiNiSi type structure (space group  $Pnma$ ), and its electronic structure was investigated by XPS measurements. The spectrum was recorded at room temperature with Al  $K\alpha$  excitation radiation (1486.6 eV), showing the two peaks for divalent Yb  $4f_{7/2}$  and  $4f_{5/2}$  at 0.5 and 1.7 eV binding energy, rendering the system purely divalent.<sup>280</sup>

#### Further Ternary Compounds

The tetrelides  $\text{YbNi}_2\text{Ge}_2$ ,  $\text{YbPd}_2\text{Si}_2$ , and  $\text{YbCu}_2\text{Si}_2$  all crystallize in the ThCr<sub>2</sub>Si<sub>2</sub> type structure (space group  $I4/mmm$ ). Their spectra show the 4f level and 4d core levels; in all of them,  $\text{Yb}^{3+}$  and  $\text{Yb}^{2+}$  are visible, and the divalent peaks are in the range of 0–2 eV for the 4f level and around 185 eV in the 4d spectra. The signal from the trivalent atoms forms a multiplet ranging from 5 to 12 eV and centered around 195 eV for the 4f and 4d spectra, respectively. Due to the presence of the two Yb valence species, these compounds are mixed valent.<sup>281</sup>

The valence of Yb in  $\text{YbInCu}_4$  was investigated by Maeda et al. in 2019 using HAXPES with an excitation energy of 8915–8965 eV at 20 and 70 K. The Yb  $3d_{5/2}$  spectra show a  $\text{Yb}^{2+}$  single peak at 1520 eV and the  $\text{Yb}^{3+}$  multiplet at 1525–1540 eV. With lower temperature, the intensity of the divalent Yb increases, while the trivalent Yb decreases. The mean valence at 70 K is +2.91 and +2.67 at 20 K.<sup>282</sup> Before this, investigation on  $\text{Yb}_{0.5}\text{In}_{0.5}\text{Cu}_2$  via XPS measurements at 20 K and excitation energies of 90 to 500 eV also showed both species present in the spectra. A valence of +2.6 was calculated from the data.<sup>276</sup> The series of  $\text{YbInCu}_4$ ,  $\text{YbCdCu}_4$ , and  $\text{YbMgCu}_4$  was investigated between 10 to 300 K. The Yb 4d XPS spectra show both Yb species in all samples. The valence of the Cd- and In-containing compounds gradually decreases with decreasing temperature, while the valence in  $\text{YbMgCu}_4$  does not change to lower temperatures.<sup>283</sup> Upon replacing In with a precious metal like Ag, Au, and Pd,  $\text{YbAgCu}_4$ ,  $\text{YbAuCu}_4$  and  $\text{YbPdCu}_4$  can be obtained. The electronic structure of these three compounds was investigated by Kang et al. The XPS spectra were recorded with Al  $K\alpha$  radiation and an excitation energy of 1486.6 eV at a temperature around 110 K. The spectra show the 4f line of Yb and the Cu 3d, Pd 4d, Pt 5d, and Ag 4d lines. Due to the overlap of the transition metal signals with the emission of divalent Yb between 0 and 4 eV, no valence can be reliably determined. However, based on the high intensity of the trivalent Yb multiplet, it can be assumed that the three compounds have a valence close to  $\text{Yb}^{3+}$  and also that the Pd containing compound is closer to the trivalent state than the Au- and Ag-containing ones.<sup>284</sup>

$\text{YbPd}_2\text{Al}_3$  and the solid solution  $\text{Yb}_{1-x}\text{Ca}_x\text{Pd}_2\text{Al}_3$  crystallize in the hexagonal  $\text{YNi}_2\text{Al}_3$  type structure ( $P6/mmm$ ), containing two crystallographically different Yb positions. The solid solution has a mixing of Yb and Ca on both Yb positions. The compounds were analyzed by XPS with Al  $K\alpha$  radiation

(1486 eV). The spectra depict the Yb 4d lines and possess five peaks for the Yb<sup>3+</sup> and two rather small peaks forming a doublet for the Yb<sup>2+</sup>, in which a partial overlap of the trivalent multiplet with the divalent Yb peaks occurs. The calcium-substituted samples Yb<sub>0.33</sub>Ca<sub>0.67</sub>Pd<sub>2</sub>Al<sub>3</sub> and Yb<sub>0.67</sub>Ca<sub>0.33</sub>Pd<sub>2</sub>Al<sub>3</sub> became oxidized due to the handling in air or due the X-ray radiation, leading to an increased Yb<sup>3+</sup> content, which cannot be used to determine the real amount of Yb<sup>2+</sup> in the samples.<sup>67</sup>

The Kondo lattice compounds Yb<sub>2</sub>Al<sub>15</sub>Pt<sub>6</sub> and Yb<sub>2</sub>Ga<sub>15</sub>Pt<sub>6</sub> were investigated via XPS at an excitation energy of 5.95 keV at 20 to 300 K. The Yb 3d levels indicate the presence of both di- and trivalent Yb. Also, in the Al-containing compound, with decreasing temperature, the intensity of the Yb<sup>3+</sup> multiplet (1524–1536 eV) decreases and the Yb<sup>2+</sup> single peak (1520 eV) intensity is observable, shifting the valence from +2.89 at 250 K to +2.83 at 20 K. In comparison, the Ga compound possesses a lower Yb valence of +2.34. However, this valence changes only minimally toward lower temperatures, meaning the Yb valence in Yb<sub>2</sub>Ga<sub>15</sub>Al<sub>6</sub> has nearly no temperature dependence. The peak of divalent Yb is at 1568 eV, and the multiplet of trivalent Yb at 1574–1582 eV.<sup>285</sup>

### Zintl Phases

The Zintl compounds Yb<sub>14</sub>MnSb<sub>11</sub> and Yb<sub>14</sub>ZnSb<sub>11</sub> (both Ca<sub>14</sub>AlSb<sub>11</sub> type, space group *I4<sub>1</sub>/acd*) were measured by Holm et al. to analyze the Yb valence by XPS analysis with an excitation energy of 1250 eV. The spectrum for the Mn-containing sample shows the Yb 4s, 4p<sub>1/2</sub>, 4p<sub>3/2</sub>, 4d<sub>3/2</sub>, and 4d<sub>5/2</sub> core levels corresponding to the binding energies of 486, 395, 345, 197, and 188 eV. For the Zn sample, the Yb levels are slightly shifted and found at 484, 394, 345, 195, and 187 eV. Additional valence band scans with an excitation energy of 120 eV were taken, showing the Yb 4f multiplet for Yb<sup>3+</sup> and the doublet of Yb<sup>2+</sup>, respectively. Also, it is stated that these compounds are prone to oxidation, leading to a higher amount of Yb<sup>3+</sup> on the surface. The amount of trivalent Yb for Yb<sub>14</sub>MnSb<sub>11</sub> is small compared to the divalent Yb and originates from the surface oxidation, in comparison to Yb<sub>14</sub>ZnSb<sub>11</sub>, which has a higher amount of Yb<sup>3+</sup>, leading to the assumption that trivalent Yb is intrinsically present in the sample, making Yb<sub>14</sub>ZnSb<sub>11</sub> an intermediate Yb valence system.<sup>286</sup>

The Yb-filled skutterudites Yb<sub>0.075</sub>CoSb<sub>3</sub>, Yb<sub>0.5</sub>CoSb<sub>3</sub>, and YbFe<sub>4</sub>Sb<sub>12</sub> have been studied by XPS measurements, showing the presence of both Yb<sup>2+</sup> and Yb<sup>3+</sup> in these compounds. The binding energies are in good agreement with Yb metal and Yb<sub>2</sub>O<sub>3</sub>, corresponding to divalent and trivalent Yb, respectively. The doublet for Yb<sup>2+</sup> is located at 1–2.2 eV, while the multiplet corresponding to Yb<sup>3+</sup> is between 6 and 14 eV.<sup>287</sup>

## MAGNETIC RESONANCE SPECTROSCOPY

This chapter is supposed to give an overview about studies in which magnetic resonance spectroscopy was applied to intermetallic Yb compounds. Nuclear magnetic resonance (NMR) in these systems can on one hand be used directly on the ytterbium nucleus to investigate the electronic situation on the Yb atom itself, while on the other hand hetero nuclear NMR is important for the measurement of effects concerning the influence of Yb on its surrounding structure.<sup>288,289</sup>

In addition, electron paramagnetic resonance (EPR) spectroscopy is a valuable option for paramagnetic species. Here most of the studies were conducted to investigate the interesting physical properties such as the Kondo effect or

heavy Fermion behavior.<sup>290–292</sup> These methods are, so to say, complementary to many other methods like, e.g., structure determination, since NMR spectroscopy is mainly used to confirm the results of the obtained structural models since it is a site-specific technique. Temperature-dependent measurements of both methods can give information about the temperature of possible phase transitions as well as the electronic situations. Here, important information about the interactions of the 4f electrons and the conduction electrons in the intermetallic compounds are obtained.

### Nuclear Magnetic Resonance (NMR) Spectroscopy

Starting with Yb NMR spectroscopy itself, it must be mentioned that in theory the application of NMR spectroscopy to two different Yb isotopes could be possible: <sup>171</sup>Yb (*S* = 1/2, 14.3% natural abundance, 7.4 MHz T<sup>-1</sup>) and <sup>173</sup>Yb (*S* = 5/2, 16.4% natural abundance, 1.3 MHz T<sup>-1</sup>).<sup>288</sup> In literature, to the best of our knowledge, except for the very first study on Yb metal, no <sup>173</sup>Yb study has been conducted, which is obviously due to the higher resonance frequency and the absence of a quadrupole moment in <sup>171</sup>Yb.<sup>288</sup>

As for all compounds with ytterbium, the successful application of magnetic resonance is limited to the diamagnetic closed-shell systems. Besides Yb, only Sc, Y, La, and Lu can play a role here. Therefore, in principle one must exclude the trivalent Yb which has a paramagnetic ground state (4f<sup>13</sup>). The magnetic moment of Yb<sup>3+</sup> leads to extremely short relaxation times and therefore to substantial line broadening and makes the detection of a NMR signal impossible.<sup>288</sup> However, examples making use of these significant facts have been reported.

#### <sup>171</sup>Yb NMR

In one of the already mentioned early studies<sup>289</sup> of <sup>171</sup>Yb NMR spectroscopy, the resonance of metallic Yb and the resonance in the intermetallic aluminide YbAl<sub>2</sub> (MgCu<sub>2</sub> type, space group *Fd3m*) were compared to other divalent ionic species, namely YbCl<sub>2</sub> (SrI<sub>2</sub> type, space group *Pbca*) and YbS (NaCl type, space group *Fm3m*). Knight shifts were determined to be around 0 ppm in the metal and about 7.76% in the intermetallic compound YbAl<sub>2</sub>. No temperature dependence of the measured Knight shift was found.<sup>289</sup>

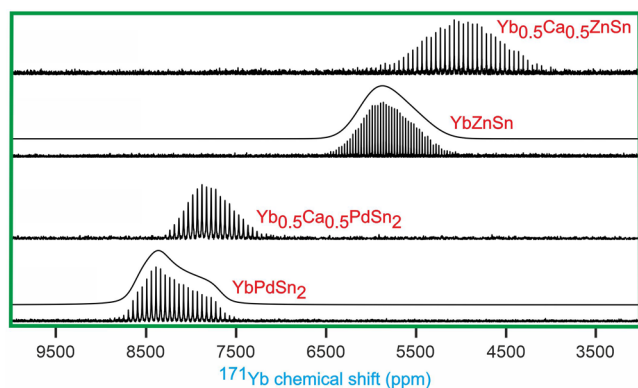
In a following study, the two aluminides YbAl<sub>2</sub> (MgCu<sub>2</sub> type, space group *Fd3m*), and YbAl<sub>3</sub> (Cu<sub>3</sub>Au type, space group *Pm3m*) as prototypes for temperature dependent valence fluctuating systems were investigated with NMR spectroscopy. Here it was possible to detect a signal for the magnetic Yb<sup>3+</sup> ion at low temperatures (1.4 to 4.2 for YbAl<sub>3</sub> and 4.2–80 K in the case of YbAl<sub>2</sub>), which is hardly the case in other studies. A determination of the hyperfine field and the relaxation rates has been carried out as well. An extremely large Knight shift of ~100% was identified in YbAl<sub>3</sub>.<sup>293</sup> In a recent <sup>27</sup>Al-NMR and Raman spectroscopic study, the partially divalent character of Yb in the cubic Laves phase YbAl<sub>2</sub> at room temperature was investigated by comparing the s electron densities with the observed isotropic resonance shifts δ(<sup>27</sup>Al). Here, Yb is not comparable to the results obtained for the formally trivalent diamagnetic members of the series.<sup>294</sup>

The Kondo semiconductor YbB<sub>12</sub> (UB<sub>12</sub> type, space group *Fm3m*) has been characterized at low temperatures using <sup>11</sup>B as well as <sup>171</sup>Yb NMR spectroscopy.<sup>295,296</sup> It was possible to observe a spin-echo signal at temperatures below 15 K for <sup>171</sup>Yb. In this case, the results of the 1/T<sub>1</sub> relaxation of <sup>171</sup>Yb is in contrast with the previous <sup>11</sup>B experiments. It is suggested

that the large susceptibility arises from magnetic moments of the  $\text{Yb}^{3+}$  ion, which stands in contrast to results obtained via  $^{11}\text{B}$  NMR spectroscopy. The effect of such a behavior is still not fully understood.<sup>295</sup> The studies discussed above show the rare case of the occurrence of a detectable NMR signal for trivalent or valence fluctuating, magnetic ytterbium. Here, the NMR signal is found at drastic shifts compared to divalent ionic Yb standards.

In agreement with the structural analysis of  $\text{Yb}_3\text{Ag}_4\text{Mg}_{12}$  ( $\text{Gd}_3\text{Ru}_4\text{Al}_{12}$  type, space group  $P6_3/mmc$ ) via X-ray diffraction (one Yb site), one single resonance for the  $^{171}\text{Yb}$  NMR spectra was observed. The temperature-dependent magnetic susceptibility measurements for this compound show the characteristics of a diamagnetic material, underlining the divalent character of Yb. The resonance of the strongly broadened spectrum recorded under static conditions applying the WURST-CPMG pulse sequence is found to be between 6800 and 7000 ppm, indicating a strong positive Knight shift. Between 200 and 300 K, the temperature dependence of the resonance shift was negligible, suggesting a strong orbital contribution to the Knight shift.<sup>99</sup>

Recently, the influence on the  $^{171}\text{Yb}$  NMR signal, when substituting ternary Yb compounds with Ca, was shown. The divalent, ternary compounds  $\text{YbZnSn}$  (NdPtSb type, space group  $P6_3mc$ ),  $\text{YbPdSn}_2$ , and  $\text{YbAuIn}_2$  (both MgCuAl<sub>2</sub> type, space group  $Cmcm$ ) as well as the solid solutions ( $\text{Yb}_{0.5}\text{Ca}_{0.5}\text{ZnSn}$ , ( $\text{Yb}_{0.5}\text{Ca}_{0.5}\text{PdSn}_2$ , and ( $\text{Yb}_{0.5}\text{Ca}_{0.5}\text{AuIn}_2$ ) were investigated with  $^{171}\text{Yb}$  NMR spectroscopy applying static WURST-CPMG pulse sequences.<sup>297</sup> For all three ternary samples, the obtained results are in excellent agreement with the structural investigations and the magnetic susceptibility data for the compounds. In all cases, a single broadened  $^{171}\text{Yb}$  spectrum could be recorded, indicating one crystallographic Yb site and underlining the diamagnetic character of  $\text{Yb}^{2+}$  in this compound. For  $\text{YbAuIn}_2$ , the magnetic susceptibility data suggest paramagnetic impurities (resulting from intermediate-valent  $\text{YbAuIn}$ ) in higher amounts than for the other compounds, which is given as the reason why here only a poor signal-to-noise-ratio was achieved in the NMR spectroscopic data. After Ca incorporation, a shift for the center of gravity as well as a line broadening is observed. This is due to the different local surrounding caused by the statistic mixing in a solid solution. As an example, the obtained NMR spectra are shown in Figure 13.



**Figure 13.** NMR spectra of  $\text{YbZnSn}$  and  $\text{YbPdSn}_2$  and the respective Ca-substituted variants  $\text{Yb}_{0.5}\text{Ca}_{0.5}\text{ZnSn}$  and  $\text{Yb}_{0.5}\text{Ca}_{0.5}\text{PdSn}_2$ .<sup>297</sup>

The germanide  $\text{YbPtGe}_2$  crystallizes in the orthorhombic  $\text{YrGe}_2$  type structure (space group  $Immm$ ) with two distinct Yb sites. Via  $^{171}\text{Yb}$  NMR spectroscopy, it was possible to prove that indeed two distinct Yb sites are present. Moreover, it becomes evident that one site is nearly divalent, while the Yb atoms on the other site are nearly trivalent. The authors explain this by the rather small shift of the observed signal. As mentioned above, in the case of paramagnetic ytterbium ions, larger Knight shifts are expected. Therefore,  $\text{YbPtGe}_2$  is named a multivalent charge-ordered compound having two different Yb valences in one compound, with no dynamic behavior.<sup>298</sup>

Finally, in the compound  $\text{YbGaGe}$  (YPtAs type,  $P6_3/mmc$ ) which has two Yb sites in an equivalent coordination, it is quite remarkable that, in a combined  $^{171}\text{Yb}$  and  $^{69,71}\text{Ga}$  NMR spectroscopic study, the two different Yb sites could be identified separately from each other in NMR spectra recorded at 4.2 K, confirming the proposed crystal structure. Here, no hints for trivalent or valence fluctuating Yb atoms could be found. The Knight shifts are reported with 0.41% (4100 ppm) and 1.1% (11000 ppm), respectively, being in the typical range of divalent nonmagnetic ytterbium compounds.<sup>299</sup>

### EPR

In contrast to the already discussed NMR spectroscopic studies on intermetallic Yb compounds, in which Yb should preferably be in the divalent state, an EPR signal should be detectable, when paramagnetic species such as the open-shell  $\text{Yb}^{3+}$  are present. Furthermore, it is possible to obtain electron spin resonance spectra which occur due to conduction electrons. Therefore, EPR spectroscopy is a method that is able to probe the local moments of unpaired 4f electrons of a lanthanide atom as well as the interaction with the surrounding delocalized conduction electrons.<sup>211</sup> By temperature-dependent measurements of the g tensor and the line width of the signal, information about the ground state of f electron systems responsible for some important phenomena can be obtained.<sup>300</sup> For the determination of the valence state of the Yb atom, the results of EPR spectroscopy can be combined with magnetic measurements. In many recent studies, EPR spectroscopy has been applied to investigate Yb compounds with exceptional low temperature physical properties, such as the Kondo effect and heavy Fermion behavior.

In Kondo systems, the EPR signal should be affected by the Kondo effect in two ways: a temperature-dependent g shift due to the local susceptibility and an additional contribution to the EPR line width.<sup>301</sup> At this point, it is interesting to note that it seems to be necessary to have ferromagnetic fluctuations in samples to observe any EPR signal, which could be shown by a study of antiferromagnetic ( $\text{CeOsPO}$ ) and ferromagnetic ( $\text{CeRuPO}$ ) samples, since only the latter give rise to an EPR signal.<sup>302</sup>

In this chapter, only some selected EPR spectroscopic studies on the most prominent examples of intermetallic Yb compounds shall be summarized. As already mentioned, this quite often goes hand-in-hand with exceptional physical properties of these compounds.

In the first part, some examples will be discussed in which the rare earth atoms were introduced as a substituent/dopant/impurity. Later, it was shown that it is possible to receive experimental EPR spectroscopic data on phase-pure unsubstituted/undoped Yb samples, so to say in a dense lattice.

In many systems, it was not possible to obtain EPR spectra and thereby information about the electronic structure of rare

earth atoms by applying EPR spectroscopy on dense lattice systems. Due to ultrafast relaxation times, a doping or substitution of paramagnetic ions into diamagnetic compounds is necessary.<sup>290</sup> One example for this is the EPR spectrum of ytterbium recorded in elemental gold (800 and 1600 ppm). Here, temperature-dependent measurements between 0.1 and 1 K resulted in the observation of a nonlinear behavior of the line width when lowering the temperature. This could be attributed to the Kondo effect.<sup>301</sup> In a study for the Yb-doped samples (100 up to 1000 ppm) with the composition  $MPd_3$  ( $M = Sc, Y, Lu, La,$  and  $Ce$ ), crystallizing in the cubic  $Cu_3Au$  structure ( $Pm\bar{3}m$ ), for all tested metals ( $Er^{3+}, Gd^{3+},$  and  $Yb^{3+}$ ), a  $\Gamma_7$  ground state indicating trivalent character of the ytterbium was found. The EPR spectrum of  $YPd_3$  (1000 ppm Yb) given in reference<sup>300</sup> is affected by the hyperfine interaction of the localized moment with the two isotopes  $^{171}Yb$  and  $^{173}Yb$  resulting in additional satellites besides the central transition.

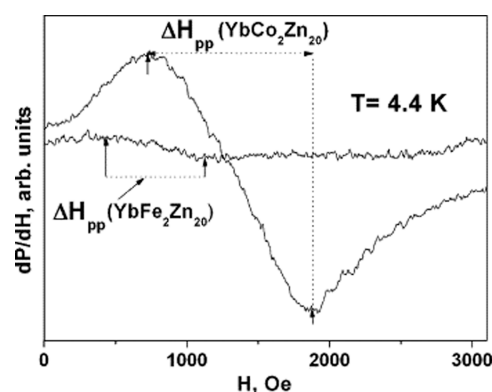
Some examples of EPR spectra of Yb in undoped nonmetallic phases such as hydride ( $YbH_x$ )<sup>303</sup> or oxide phases ( $Yb_xY_{1-x}Ba_2Cu_3O_y$ )<sup>304</sup> are reported. In these systems, a narrowing of the observed EPR line is described because of the spin-glass ordering and the absence of heavy Fermion behavior as well as exchange narrowing. In an early example, it is reported that the temperature-dependent EPR spectra of pure  $YbBe_{13}$  and substituted diamagnetic  $Y_{1-x}Yb_xBe_{13}$  ( $x = 0.01$  and  $0.09$ ) are identical, which is linked to the rather large separation of  $\sim 500$  pm of the rare earth atoms in the  $NaZn_{13}$  type structure ( $Fm\bar{3}c$ ), leading to a small crystal field splitting ( $\Gamma_7 \rightarrow \Gamma_6 \rightarrow \Gamma_8$ ) and next to no interaction between the magnetic moments except for some RKKY exchange. For the exact CEF, the differences in the literature were already discussed above. In this EPR spectroscopic study, the valence state of Yb was determined to be trivalent with a  $\Gamma_7$  ground state, equivalent to Eu and Sm, and in contrast to Ce which is in an intermediate valence state in this structure type.<sup>236</sup>

When looking in the literature, the application of EPR spectroscopy to intermetallic Yb compounds showing heavy Fermion behavior is the most prominent one. Especially, the compound  $YbRh_2Si_2$  was the first compound in which the direct investigation of the EPR signal of the rare earth ion responsible for the Kondo effect, up to 25 K, so below and above the Kondo temperature, without any doping, substitution, or impurities, was possible. The detection of the signal was thought to be impossible until this proof. In previous studies of doped samples (e.g., with  $Gd^{3+}$ ), one cannot exclude the influence of the dopant on the Kondo lattice. The exact origin of the EPR signal in this dense Kondo lattice below the relatively high Kondo temperature of  $T_K = 20$  K in this compound is yet unclear.<sup>158,305,306</sup> A similar EPR signal could be detected a little later in the isostructural silicide  $YbIr_2Si_2$ . The results are in good agreement with those of the Rh compound. Way below the Kondo temperature of 40 K, it was possible to detect a strong EPR signal down to 1 K. Therefore, it is the second example of strong, intense EPR lines in a dense Kondo heavy Fermion system.<sup>307</sup> In the past years, more examples of direct EPR detection in heavy Fermion compounds were reported. For  $YbNiAl_2$ , the trivalent state of the Yb atom was confirmed by measuring an intense EPR spectrum below 20 K. However, here the authors mention that the intensity of the signal obtained at temperatures below 20 K was about an order of magnitude lower than for  $YbRh_2Si_2$ . By increasing the temperature, the EPR signal disappeared. The reason for the appearance of the EPR signal at low

temperatures in this dense heavy Fermion compound is not only the already mentioned ferromagnetic fluctuations but also magnetic anisotropy as well as cumulative effects of the surrounding crystal field and spin–orbit interactions.<sup>308</sup>

For the case of the above-mentioned heavy Fermion superconductor  $\beta$ - $YbAlB_4$ , an interesting dual nature behavior of the EPR signal was observed. At high temperatures, an EPR signal was found for the Yb as well as for the Pauli-paramagnetic Lu compound. The Lu compound shows the characteristic temperature dependence of a conduction electron spin resonance over the whole temperature range (4.2 to 295 K), while in comparison, at 4.2 K, the Yb compound shows a hyperfine splitting caused by the  $^{171}Yb$  isotope, confirming the characteristics of an  $Yb^{3+}$ -caused spin resonance. This is a unique behavior of the dual nature observed in EPR spectroscopy.<sup>211</sup>

For the compounds  $YbBiPt$  and  $YbRh_2Pb$ , EPR measurements could be performed at temperatures of up to 110 K, although for discussion of the nature of the compounds, measurements at 4.4 K were used. The results give hints about the crystal electric field (CEF) for the  $Yb^{3+}$  atom in these ternary compounds. For the plumbide, an extremely small hybridization of the 4f electrons and the conduction electrons is suggested together with weak RKKY interactions. In contrast, the bismuthide shows a strongly broadened CEF due to large hybridization effects.<sup>290</sup> In this context, one must mention one important class of isostructural compounds. All Yb compounds with the general formula  $YbT_2Zn_{20}$  ( $T = Fe, Co, Ru, Rh, Os, Ir$ )<sup>309</sup> exhibit heavy Fermion behavior. Therefore, a comparison of the spectroscopic properties such as for the above-mentioned silicides  $Yb(Ir,Rh)_2Si_2$  was possible.<sup>309</sup> Here, the Co and Fe compounds have been investigated with EPR spectroscopy. In an analogous way to the already mentioned studies, it was possible to detect a spectrum without any dilution or paramagnetic impurity. A direct investigation of the dense Yb lattice was possible.<sup>290,305,309</sup> A comparison of the EPR absorption of the Co and Fe compounds (Figure 14) gave hints about the



**Figure 14.** EPR spectra of  $YbCo_2Zn_{20}$  and  $YbFe_2Zn_{20}$ . Reprinted from ref 290 with permission from Elsevier.

involvement of the d electrons of the magnetic atoms. With the Co compound EPR absorption being higher by a factor of 10, it was proposed that the increasing number of available d electrons also plays a crucial role in the intensity of the EPR signal and not only the localized magnetic moment of the f electrons on the rare earth atom.



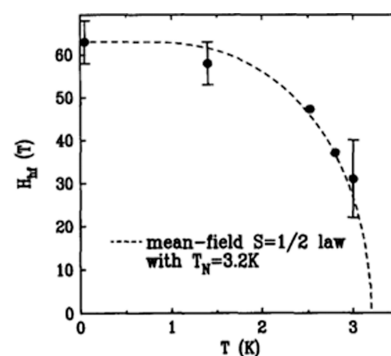
<sup>170</sup>Yb MÖSSBAUER SPECTROSCOPY

<sup>170</sup>Yb is a Mössbauer-active isotope and thus allows studies of the ytterbium valence state, the coordination geometry, and magnetic hyperfine field splitting. The energy  $E_\gamma$  amounts to 84.25 keV. The decisive point for <sup>170</sup>Yb Mössbauer spectroscopic studies is the very low natural abundance of <sup>170</sup>Yb of only 3%, forcing long data collection times. The basic nuclear moment data for the different ytterbium isotopes (<sup>170</sup>Yb, <sup>171</sup>Yb, <sup>172</sup>Yb, <sup>174</sup>Yb, <sup>176</sup>Yb) have been summarized by Stevens and Dunlap;<sup>310</sup> however, <sup>170</sup>Yb is the usually studied isotope. Its basic Mössbauer spectroscopic characteristics are:  $I_c = 2$  (nuclear spin operator in the excited state);  $I_g = 0$  (nuclear spin quantum number in the ground state); transition energy of 84.25 keV;  $TmB_{12}$  as the usual source (typically 20 mCi) with a half-life of 130 d; and natural line width of 2.03 mm s<sup>-1</sup>.<sup>288,311–313</sup>

The <sup>170</sup>Yb isomer shift values for divalent and trivalent ytterbium vary only over a small range. For purely ionic compounds, representative values are  $\delta = -0.49$  mm s<sup>-1</sup> for Yb<sup>II</sup>SO<sub>4</sub> and +0.09 mm s<sup>-1</sup> for Yb<sup>III</sup>Cl<sub>3</sub>.<sup>311</sup> It is clear that this marginal separation of the Yb<sup>II</sup> and Yb<sup>III</sup> isomer shifts hampers the studies of valence changes.<sup>314</sup> However, the <sup>170</sup>Yb data is very sensitive to the hyperfine splitting of the nuclear energy levels and thus allows detailed studies of the magnetic ordering phenomena. The basics for the <sup>170</sup>Yb Mössbauer spectroscopic data with respect to magnetic hyperfine splitting are summarized in a concise resource letter by Cadogan and Ryan.<sup>312</sup> In the large field of ytterbium intermetallics, the magnetic ordering of the trivalent compounds often sets in only at very low temperatures. Consequently, <sup>170</sup>Yb Mössbauer spectra have frequently been measured down to the mK range. Determination of the quadrupole splitting parameter is the important feature for discriminating divalent and trivalent ytterbium. Yb<sup>II</sup> has a filled 4f shell and thus no 4f contribution to the electric field gradient at the <sup>170</sup>Yb nucleus. The Yb<sup>III</sup> compounds, in contrast, show paramagnetic quadrupole triplets.<sup>315</sup> Typical examples for divalent ytterbium intermetallics are the binary ytterbium–cadmium phases YbCd, YbCd<sub>2</sub>, Yb<sub>14</sub>Cd<sub>51</sub>, YbCd<sub>5,7</sub>, and YbCd<sub>6</sub><sup>316</sup> or the ternary stannide Yb<sub>3</sub>Pd<sub>2</sub>Sn<sub>2</sub>.<sup>317,318</sup>

The magnetically ordered Yb<sup>III</sup> compounds show more or less well-resolved hyperfine field splitting. From the hyperfine field value  $H_{hf}$ , one can derive the ordered moment  $\mu$  through the relation  $H_{hf} = C\mu$  with a value of  $C = 102$  T/ $\mu_B$  for Yb<sup>3+</sup> (derived from accurate measurements of the paramagnetic hyperfine constant of <sup>170</sup>Yb<sup>3+</sup> in the cubic  $\Gamma_7$  state).<sup>319–321</sup> Due to the meaninglessness of the isomer shift value, the <sup>170</sup>Yb Mössbauer spectroscopic studies on intermetallic compounds only report on the quadrupole splitting parameters and the magnetic hyperfine fields.

Temperature-dependent measurements in the magnetically ordered regimes allow for monitoring the evolution of the magnetic hyperfine field with decreasing temperature. These data can be fitted with an  $S = 1/2$  law, and the magnetic ordering temperature can be derived as complementary experiment to the magnetic characterization. Data of binary Yb<sub>3</sub>Pd<sub>4</sub><sup>322</sup> is presented in Figure 15. Further related examples are YbNiBC,<sup>323</sup> or samples from the solid solution Yb<sub>5</sub>Si<sub>x</sub>Ge<sub>4-x</sub>.<sup>324</sup> In several cases, the crystal field properties have also been derived from the <sup>170</sup>Yb Mössbauer spectroscopic data and compared to  $L_{III}$ -edge X-ray absorption spectra



**Figure 15.** Temperature dependence of the hyperfine field at the <sup>170</sup>Yb nuclei in Yb<sub>3</sub>Pd<sub>4</sub> ( $T_N = 3.2$  K). The thermal variation follows an  $S = 1/2$  mean field law (dashed line). Reprinted from ref 322 with permission from Elsevier.

and magnetic data. An exemplary investigation is the <sup>170</sup>Yb Mössbauer spectroscopic characterization of YbCu<sub>3</sub>Al<sub>2</sub>.<sup>325</sup>

Early work on <sup>170</sup>Yb Mössbauer spectroscopic data focused on the rocksalt type monopnictides YbP, YbAs, and YbSb (space group  $Fm\bar{3}m$ ),<sup>326–328</sup> which are Kondo-frustrated antiferromagnets with very low Néel temperatures <1 K (Table 3). Although the three phases have cubic symmetry, inhomogeneous hyperfine fields/hyperfine field distributions have been observed. Thus, <sup>170</sup>Yb Mössbauer spectroscopy is an excellent complementary tool for monitoring structural distortions. In the following, many other binary, ternary, and quaternary intermetallic phases have been characterized through their <sup>170</sup>Yb Mössbauer spectra. Some selected phases, which all have their own peculiarity, are shortly discussed in the following.

The incommensurate modulated structure of YbPtAl (TiNiSi type, space group  $Pnma$ , wave-vector  $k = 0.30, 0, 0$ ) was studied by a combination of neutron powder diffraction and <sup>170</sup>Yb Mössbauer spectroscopic data.<sup>329</sup> The low-temperature spectroscopic data (0.025 K) clearly shows that the modulated structure is not an antiphase one. The maximum moment of 2.5  $\mu_B$  was derived from the 1.6 K spectrum.

The <sup>170</sup>Yb Mössbauer spectroscopic data of YbNiAl<sub>4</sub> (YNiAl<sub>4</sub> type, space group  $Cmcm$ ) is an important complementary result to specific heat data. While the latter points to the onset of magnetic ordering, no hyperfine field splitting is evident from the <sup>170</sup>Yb Mössbauer spectra down to 1.5 K.<sup>23</sup> This is consistent with an intermediate ytterbium valence, since the temperature dependence of the magnetic susceptibility also revealed a substantially reduced experimental magnetic moment.

The ytterbium valence change induced by substitution was studied for the solid solutions Yb<sub>5</sub>Ge<sub>4-x</sub>(Sb,Ga)<sub>x</sub> with  $x \leq 1$  (Sm<sub>5</sub>Ge<sub>4</sub> type, space group  $Pnma$ ).<sup>330</sup> For influencing the ytterbium valence, the germanium atoms are partially substituted by gallium with lower and antimony with higher electron count. The Yb<sup>II</sup>/Yb<sup>III</sup> ratio could be derived from well-resolved <sup>170</sup>Yb Mössbauer spectra at 5 K through the subsignals with and without quadrupole splitting (vide infra). The Mössbauer spectroscopic data show an almost gradual increase of the Yb<sup>II</sup>/Yb<sup>III</sup> ratio from ~0.95 for Yb<sub>5</sub>Ge<sub>3</sub>Ga to ~1.55 for Yb<sub>5</sub>Ge<sub>3</sub>Sb.

The quadrupolar phase transition at 80 K in YbAl<sub>3</sub>C<sub>3</sub> (ScAl<sub>3</sub>C<sub>3</sub> type, space group  $P6_3/mmc$ ) was monitored by <sup>170</sup>Yb Mössbauer spectroscopy.<sup>331</sup> Spectra were collected in the

Table 3.  $^{170}\text{Yb}$  Mössbauer Spectroscopy Derived Parameters for Selected Intermetallic Ytterbium Compounds<sup>a</sup>

Compound	Structure Type	$T_M/\text{K}$	$T_{MB}/\text{K}$	$B_{\text{Hf}}/\text{T}$	$\mu/\mu_B$	Reference
$\text{YbBe}_{13}$	$\text{NaZn}_{13}$	$T_N = 1.28$	0.05	172	1.73	336
$\text{YbB}_{12}$	$\text{UB}_{12}$	—	0.04	<10	<0.1	337
$\text{YbP}$	$\text{NaCl}$	$T_N = 0.41$	0.045	81	0.79	326
$\text{YbAs}$	$\text{NaCl}$	$T_N = 0.58$	0.045	81	0.82	326
$\text{YbSb}$	$\text{NaCl}$	$T_N = 0.32$	0.045	64	0.63	326
$\text{YbNi}_5$	$\text{CaCu}_5$	$T_C = 0.55$	0.05	399(2)	3.9	338
$\text{Yb}_3\text{Pd}_4$	$\text{Pu}_3\text{Pd}_4$	$T_N = 3.2(1)$	0.05	63(5)	0.62(5)	322
$\text{YbCu}_3\text{Al}_2$	$\text{PrNi}_2\text{Al}_3$	$T_N = 1.95$	0.12	290(10)	2.84(10)	325
$\text{YbPtAl}$	$\text{TiNiSi}$	$T_N = 5.8$	4.2	159	1.56	329
$\text{Yb}_3\text{Cu}_4\text{Ge}_4$	$\text{Gd}_3\text{Cu}_4\text{Ge}_4$	$T_N = 7.5$	1.45	370	3.6	332
				148	1.45	
$\text{Yb}_2\text{Co}_3\text{Al}_9$	$\text{Y}_2\text{Co}_3\text{Al}_9$	$T_N = 1.2$	0.06	95	0.93	333
$\text{YbNiSb}$	$\text{MgAgAs}$	$T_N = 0.85$	0.13	108(10)	1.06(10)	334
$\text{YbNiSn}$	$\text{TiNiSi}$	$T_N = 5.65$	1.48	870(20)	0.85(2)	125

<sup>a</sup>Parameters include: magnetic ordering temperature,  $T_M$ ; measurement temperature for the Mössbauer spectrum,  $T_{MB}$ ; magnetic hyperfine field,  $B_{\text{Hf}}$ ; and magnetic moment at the  $^{170}\text{Yb}$  nuclei,  $\mu$ . Note that  $\text{Yb}_3\text{Cu}_4\text{Ge}_4$  shows two sub-spectra with different hyperfine fields. Standard deviations are given in parentheses, when available.

entire range from 0.04 to 150 K. No magnetic ordering of the  $\text{Yb}^{3+}$  moments was evident down to the lowest measurement temperature. The symmetry reduction (loss of the mirror plane perpendicular to the  $6_3$  screw axis) is well expressed in the course of the quadrupole splitting in the  $^{170}\text{Yb}$  spectra in the sequence 85 K  $\rightarrow$  80 K  $\rightarrow$  77 K  $\rightarrow$  75 K with a nonaxial quadrupolar hyperfine interaction in the low-temperature phase. Also in this case,  $^{170}\text{Yb}$  Mössbauer spectroscopy served as a valuable complementary probe for studying a structural phase transition.

$\text{Yb}_3\text{Cu}_4\text{Ge}_4$  with  $\text{Gd}_3\text{Cu}_4\text{Ge}_4$  type structure (space group  $Immm$ ) is one of the remarkable ytterbium intermetallics, since it exhibits two crystallographically independent ytterbium sites  $4e$  and  $2d$ .<sup>332</sup> Although the magnetic data indicate a single transition temperature into the magnetically ordered state, these two ytterbium sites show distinctly different magnetic hyperfine spectra (Figure 16) with ordered moments of  $3.6 \mu_B$  for the  $4e$  and  $1.45 \mu_B$  for the  $2d$  site. Integration of the subspectra is consistent with the 2:1 multiplicity of the individual Wyckoff positions.

$^{170}\text{Yb}$  Mössbauer spectroscopy was employed to differentiate the magnetic behavior of the Yb atoms in the isotopic compounds  $\text{Yb}_2\text{Co}_3\text{Al}_9$  and  $\text{Yb}_2\text{Co}_3\text{Ga}_9$  ( $\text{Y}_2\text{Co}_3\text{Ga}_9$  type, space group  $Cmcm$ ).<sup>333</sup> The aluminum compound shows a stable  $^2F_{7/2}$  ground state and well-resolved hyperfine field splitting (95 T) at 0.06 K, while  $\text{Yb}_2\text{Co}_3\text{Ga}_9$  is a paramagnetic heavy-Fermion material showing no hyperfine field splitting over the whole temperature range.

The solid solution  $\text{Yb}_{1-x}\text{In}_x\text{Cu}_2$  is structurally derived from the cubic Laves phase (space group  $Fd\bar{3}m$ , vide ultra). The sample with the composition  $\text{Yb}_{0.4}\text{In}_{0.6}\text{Cu}_2$  shows a sharp valence transition around 50 K.  $^{170}\text{Yb}$  Mössbauer spectra at 4.2 K indicate that the  $\text{Yb}^{3+}$  ions exhibit no magnetic hyperfine field splitting in the cubic environment; however, it shows a strong  $4f-5d$  hybridization.<sup>191</sup>

The final example shows the necessity of complementary techniques for the precise understanding of structure–property relations. The half-Heusler phase  $\text{YbNiSb}$  ( $\text{MgAgAs}$  type, space group  $F\bar{4}3m$ ) has been characterized through  $^{170}\text{Yb}$  Mössbauer spectra at 0.13 and 1.4 K.<sup>334</sup> Spectra in the magnetically ordered regime revealed enlarged and inhomogeneous line widths. Based on these results, it was argued that the

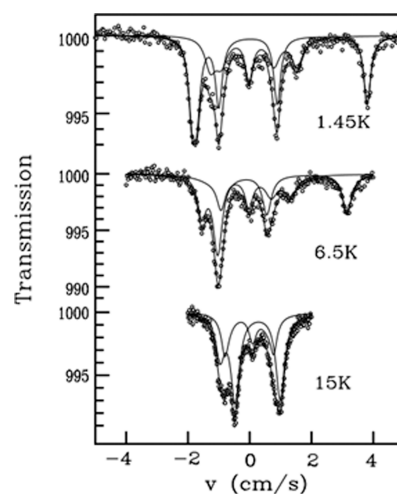


Figure 16.  $^{170}\text{Yb}$  Mössbauer spectra of  $\text{Yb}_3\text{Cu}_4\text{Ge}_4$ . The spectrum at 15 K was taken in the paramagnetic range, while the 1.45 and 6.5 K spectra are derived from the magnetically ordered phase. The subspectra are marked by different solid lines. Reprinted from ref 332 with permission from Elsevier.

quadrupolar hyperfine interaction of the paramagnetic phase has no cubic site symmetry. Later, precise single crystal X-ray diffraction data showed that  $\text{YbNiSb}$  samples are a mixture of equiatomic  $\text{YbNiSb}$  as the main component along with a distribution of  $\text{YbNi}_x\text{Sb}$  domains with different nickel content.<sup>335</sup> Thus, the magnetically split spectrum is a superposition of the magnetically split subspectra of the distribution of  $\text{YbNi}_x\text{Sb}$  domains.

Many other interesting examples for  $^{170}\text{Yb}$  Mössbauer spectroscopic studies on ytterbium intermetallics have been published. In such a review, one can only present selected examples. Besides the data discussed above, we have added some further examples in Table 3 with respect to their basic Mössbauer spectroscopic data.

Besides the many  $^{170}\text{Yb}$  Mössbauer spectroscopic studies under ambient conditions, some investigations were carried out under application of external magnetic fields or high pressure. The heavy Fermion silicide  $\text{YbRh}_2\text{Si}_2$  was additionally studied by  $^{170}\text{Yb}$  Mössbauer spectroscopy as a complementary

tool to ESR spectroscopy.<sup>339</sup> The ESR data for this silicide gave hint for a slow relaxation rate. This issue could be clarified by <sup>170</sup>Yb Mössbauer spectroscopy. The <sup>170</sup>Yb spectrum only shows a single signal, which does not match with a simulated spectrum for slow relaxation at an applied field of 0.4 T. The <sup>170</sup>Yb data thus indicate that the Yb<sup>3+</sup> ions relax more rapidly than expected from the ESR data. YbRh<sub>2</sub>Si<sub>2</sub> was furthermore studied at 1.3 K and pressures ranging from 6.4 to 16.5 GPa.<sup>340</sup> The experiments show a continuous line broadening with increasing pressure. The pressure-induced first-order magnetic phase transition sets in around 10 GPa. The line broadening indicates increasing magnetic hyperfine field splitting and thus an increase of the ordered moment at the ytterbium nuclei with a value of 1.9  $\mu_B$  ( $\equiv 191$  T) at 16.5 GPa, still reduced when compared with the free ion value of 4  $\mu_B$ .<sup>105</sup>

The technical details for <sup>170</sup>Yb Mössbauer spectroscopy under high-pressure conditions are well documented in an article by Schöppner et al.,<sup>341</sup> which focuses on the pressure-induced ytterbium valence change in YbCuAl (ZrNiAl type, space group  $P\bar{6}2m$ ). The compound was studied at 4.2 K from 0 to 130 kbar pressure. The <sup>170</sup>Yb spectra show a continuous increase of the quadrupole splitting parameter with increasing pressure, manifesting the valence transition toward trivalent ytterbium with 4f<sup>3</sup> configuration. This ground state is achieved already at around 50 kbar.

<sup>170</sup>Yb Mössbauer spectroscopic data was collected for the TiNiSi type stannide YbNiSn (space group  $Pnma$ ) up to 7 GPa.<sup>342</sup> A remarkable feature is the increase of the Curie temperature with increasing pressure with a maximum of  $T_C = 7.6$  K at 1.7 GPa followed by a rapid decrease. In parallel, the quadrupole splitting parameter and the magnetic hyperfine field increase from ambient pressure to 7 GPa. The maximum field of 175 T corresponds to an ordered moment of 1.72  $\mu_B$ .

Also, the heavy Fermion compound Yb<sub>2</sub>Ni<sub>2</sub>Al ( $W_2CoB_2$  type, space group  $Immm$ ) shows a pressure-driven first-order magnetic phase transition.<sup>343</sup> A crossover from a nonmagnetic heavy Fermion state to a magnetically ordered state occurs at pressures >8 GPa and below 2 K. The <sup>170</sup>Yb Mössbauer spectrum (1.8 K data) at the highest pressure of 9 GPa showed an ordered moment of  $\approx 1$   $\mu_B$  at the ytterbium nuclei.

Finally, we turn toward an interesting extension of <sup>174</sup>Yb Mössbauer spectroscopy which was realized recently using synchrotron radiation as the Mössbauer spectroscopy source. The classical experiments all used the <sup>170</sup>Yb isotope. The main disadvantage is its low natural abundance of only 3% as compared to 31.8% for <sup>174</sup>Yb. The synchrotron radiation-based technique is described in a resource article by Masuda et al.,<sup>344</sup> and exemplified for an YbB<sub>12</sub> ( $UB_{12}$  type, space group  $Fm\bar{3}m$ ) sample. Further advantages of this new technique are the higher brilliance of the synchrotron radiation as compared to X-rays deriving from a radioactive isotope, and there is no need for isotope enrichment. Along with the use of a windowless avalanche photodiode detector, fast data collections are possible, and this technique will also be helpful for future high-pressure studies. Two further examples for the application of <sup>174</sup>Yb Mössbauer spectroscopy are Yb<sub>3</sub>Fe<sub>14</sub>B ( $Nd_2Fe_{14}B$  type, space group  $P4_2/mnm$ )<sup>345</sup> and  $\alpha$ -YbAlB<sub>4</sub> ( $YCrB_4$  type, space group  $Pbam$ ).<sup>346</sup>

## ■ THERMOELECTRICS

Thermoelectric materials are usually heavily doped semiconductors which can produce a charge flow when exposed to a temperature gradient. Depending on the charge carrier type

in the respective material, these are denoted as p- or n-type semiconductors. The quality of a thermoelectric material is usually given by the dimensionless figure-of-merit  $zT$  which basically incorporates the Seebeck coefficient as well as the thermal and electrical conductivity. Details regarding these values and their interplay can be found in a number of reviews.<sup>347–352</sup> However, the determination of the Seebeck coefficient can also be used for identifying magnetic transitions and other physical phenomena as it is very susceptible to changes in the electronic states. In the following chapter, examples of classical intermetallic Yb-compounds as well as examples of Zintl phases, being the more typical thermoelectric materials, will be addressed.

## Binary Compounds

We will start this chapter with selected examples of classical intermetallic materials. For this, the binaries YbAl<sub>3</sub>, YbSi<sub>2</sub>, YbB<sub>6</sub>, YbCd<sub>6</sub>, and Yb<sub>3</sub>Si<sub>5</sub> as well as the solid solutions Yb<sub>1-x</sub>Sc<sub>x</sub>Al<sub>2</sub>, Yb<sub>1-x</sub>In<sub>x</sub>Cu<sub>2</sub>, and YbCu<sub>5-x</sub>Ag<sub>x</sub> have been chosen. From the ternary systems, equiatomic YbPtBi, YbNiSn, and YbPtGe, as well as YbAlB<sub>4</sub>, YbNiAl<sub>4</sub>, YbRh<sub>2</sub>Al<sub>20</sub>, and YbT<sub>2</sub>Zn<sub>20</sub> ( $T = Fe, Co, Ru, Rh, Os, Ir$ ), Yb<sub>2</sub>Pt<sub>3</sub>Sn<sub>5</sub>, Yb<sub>2</sub>T<sub>3</sub>X<sub>9</sub> ( $T = Rh, Ir; X = Al, Ga$ ), Yb<sub>2</sub>Pt<sub>6</sub>Al<sub>15</sub>, and Yb<sub>2</sub>Ni<sub>12</sub>P<sub>7</sub> were selected.

YbAl<sub>3</sub> ( $Cu_3Au$  type, space group  $Pm\bar{3}m$ ) is, as mentioned before, a well studied compound due to the ambivalent nature of the Yb atoms leading to a change of the Yb valence with temperature.<sup>113</sup> Due to this phenomenon, the Seebeck coefficient of intermediate valent/valence fluctuating Yb compounds is often significantly higher than expected. This is also the case for YbAl<sub>3</sub> where the Seebeck coefficient reaches values over  $-80$   $\mu V K^{-1}$  near 120 K,<sup>353</sup> and YbSi<sub>2</sub> ( $AlB_2$  type, space group  $P6/mmm$ ),<sup>354</sup> also a mixed valent compound, shows values between  $-50$  and  $-35$   $\mu V K^{-1}$  between 300 and 525 K. The electrical resistivity, however, possesses abnormally large values due to spin-disorder scattering. YbSi<sub>2</sub> shows an almost temperature-independent electrical resistivity and a total thermal conductivity between 8 and 11  $W m^{-1} K^{-1}$ , leading to a  $zT$  between 0.08 at 300 K and 0.04 at 525 K. YbB<sub>6</sub> adopts the cubic CaB<sub>6</sub> type structure (space group  $Pm\bar{3}m$ ) and exhibits a certain phase width between  $x = 5.7$  and 6.3.<sup>355</sup> Therefore, this compound can be used to study and tune the thermoelectric properties. The compounds with  $x = 5.7, 5.9,$  and 6.0 exhibit a negative Seebeck coefficient of approximately  $-100$   $\mu V K^{-1}$  between 300 and 1050 K. In contrast, the samples with  $x = 6.1$  and 6.3 start with a positive Seebeck coefficient of  $+100$   $\mu V K^{-1}$ , which gradually changes to negative values at higher temperatures. YbCd<sub>6</sub>, despite the same sum formula, crystallizes in a totally different crystal structure ( $YCd_6$  type, space group  $Im\bar{3}$ ). Instead of octahedral B<sub>6</sub> clusters, here a quasi-crystal (QC) approximant with a shell-like structure is present. Both the QC YbCd<sub>5.7</sub> and the approximant YbCd<sub>6</sub> were investigated with respect to their thermoelectric properties. Both show small positive Seebeck coefficients between  $+10$  and  $+20$   $\mu V K^{-1}$ . The striking feature is a maximum in the electrical resistivity around 70 K for YbCd<sub>5.7</sub>. A similar trend can be seen for the different compositions of the quasi-crystals, leading to  $zT$  values of  $5 \times 10^{-3}$  near RT.<sup>356,357</sup> Finally, Yb<sub>3</sub>Si<sub>3</sub> ( $Mn_3Si_3$  type, space group  $P6_3/mcm$ ) should be discussed as the last binary representative. Again, this compound is valence instable and shows fluctuations between di- and trivalent Yb, leading to two sign changes in the Seebeck coefficient. Below  $\sim 40$  K, it is negative with a rather sharp minimum at 10 K, followed by a

broad maximum with positive values ( $\sim 13 \mu\text{V K}^{-1}$ ) around 100 K, and again a change to a negative Seebeck coefficient close to 300 K. The resistivity is, compared to the close-shell system  $\text{Lu}_5\text{Si}_3$ , enhanced by a factor of close to two, caused again by the valence fluctuation.<sup>358</sup>

The solid solution  $\text{Yb}_{1-x}\text{Sc}_x\text{Al}_2$  (MgCu<sub>2</sub> type, space group  $F\bar{d}3m$ ) was studied to probe the influence of “chemical pressure” on the properties of the valence fluctuating system  $\text{YbAl}_2$ . Different compositions with  $x = 0.6-1$  were synthesized and investigated. Magnetic susceptibility measurements clearly indicate that the intermediate valence (IV) state gets more pronounced with increasing Sc content and a shift toward more  $\text{Yb}^{3+}$  is visible. Samples with a pronounced IV state at the same time exhibit a drastically negative Seebeck coefficient up to  $-50 \mu\text{V K}^{-1}$  below 50 K for  $x = 0.9$ . This leads to a maximum in the power factor ( $S^2/\rho$ ) of  $160 \mu\text{W cm}^{-1} \text{K}^{-2}$  for  $x = 0.9$  around 50 K.<sup>359</sup>

Since  $\text{Yb}_{1-x}\text{In}_x\text{Cu}_2$  (MgCu<sub>2</sub> type, space group  $F\bar{d}3m$ ) shows a temperature-induced first-order valence phase transition, its Seebeck coefficient was investigated for  $x = 0.6$  to probe the fine structure of the conduction electron band closely associated with the magnetic state. The temperature dependence of the Seebeck coefficient starts with a positive sign at high temperatures and exhibits a change of sign near 200 K. Around 40 K, a minimum with  $S = -16 \mu\text{V K}^{-1}$  is visible, coinciding with the temperature of the valence phase transition.<sup>360</sup>

### Equiatomic Ternary Compounds

For the ternary compounds, we will start with selected equiatomic representatives, i.e.,  $\text{YbNiSn}$ ,  $\text{YbPdGe}$ ,  $\text{YbPtGe}$ , and  $\text{YbPtBi}$ . For  $\text{YbNiSn}$  (TiNiSi type, space group  $Pnma$ ), the thermoelectric properties were studied to obtain further information about the crystal field splitting. Inelastic neutron scattering should yield three crystal field excitations based on the crystal structure. However, only two excitations were visible; therefore, the thermoelectric properties were determined to gain a deeper insight. The temperature dependence of the Seebeck coefficient shows a broad minimum around 100 K along with a peak at 7.5 K and a change of sign below 5 K. From these investigations, the degeneracy of two crystal field excitations was deduced.<sup>361</sup>  $\text{YbPdGe}$  (YbAuSn type, space group  $Imm2$ ) and  $\text{YbPtGe}$  (TiNiSi type, space group  $Pnma$ ) show a very similar behavior. The Seebeck coefficient shows a broad maximum around 120 K followed by a maximum at 30 K. Three contributions were deduced: a Mott's diffusion term as well as a Kondo and a resonance term. While the first one is small and temperature-independent, the Kondo term originates from Kondo scattering of conduction electrons by the 4f spins. The inelastic contribution to the resonance term originates from inelastic scattering of the conduction electrons by magnetic interactive spins, respectively.<sup>362</sup> Finally,  $\text{YbPtBi}$  (MgAgAs type, space group  $F\bar{4}3m$ ) was investigated. This compound is of particular interest since it exhibits one of the largest Sommerfeld coefficients known ( $\gamma = 7.4 \text{ J mol}^{-1} \text{K}^{-2}$ ).<sup>137</sup> Thin films of  $\text{YbPtBi}$  exhibit also a negative Seebeck coefficient with  $-54 \mu\text{V K}^{-1}$ .<sup>363</sup>

### Further Ternary Compounds

While the silicides and germanides adopt the tetragonal body centered  $\text{ThCr}_2\text{Si}_2$  type structure (space group  $I4/mmm$ ), the pnictide-based Zintl phases of the same chemical composition crystallize in the trigonal space group  $P\bar{3}m1$  and adopt the  $\text{Ce}_2\text{O}_2\text{S}$  type structure.

$\text{YbRh}_2\text{Si}_2$  is an antiferromagnetic heavy Fermion compound;<sup>162,163,364,365</sup> therefore, investigations of the thermoelectric properties (TEP) can give important information on the behavior of the conduction electrons. The TEP contain information about the relaxation time and the DOS of the quasiparticles at the Fermi level  $E_F$ . It is interesting to note that at very low temperatures, the TEP is commonly positive for Ce heavy Fermion compounds, in agreement with the Kondo picture, while it is negative for Yb heavy Fermion compounds.<sup>366</sup> Field-dependent measurements of the Seebeck coefficient  $S(H)$  have been conducted down to 120 mK and up to 16 T.<sup>367</sup> Above 3.5 T, a cascade of sharp anomalies can be observed, pointing toward topological transitions of the Fermi surface. Upon substitution of rhodium by cobalt according to  $\text{Yb}(\text{Rh}_{1-x}\text{Co}_x)_2\text{Si}_2$ ,<sup>368,369</sup> the Kondo temperature decreases significantly, and the sole large minimum in the thermopower that is observed for  $\text{YbRh}_2\text{Si}_2$  splits into two minima. In addition, the absolute thermopower is strongly reduced with increasing  $x$ , in line with a weaker exchange coupling of the 4f and the conduction electron states making pure  $\text{YbCo}_2\text{Si}_2$  a stable trivalent system.<sup>241,242,369</sup> However, still two minima are observed in the Seebeck coefficient, indicating weak residual Kondo scattering in line with a tiny residual hybridization between 4f and conduction electron states. A comparative study of the thermoelectric response under magnetic field of  $\text{YbRh}_2\text{Si}_2$  and  $\text{YbAlB}_4$  showed a significant difference in the Seebeck coefficient near the quantum critical points of both compounds. When approaching the critical field,  $S/T$  is drastically reduced in  $\text{YbRh}_2\text{Si}_2$ ; however, an enhancement in  $\beta\text{-YbAlB}_4$  is observed.<sup>370</sup>

$\text{YbPd}_2\text{Si}_2$  is another heavy Fermion material within the group of Yb intermetallics and has been studied broadly.<sup>371-373</sup> In the range between 1.5 and 300 K, the Seebeck coefficient is negative with a minimum around 50 K, which was attributed to the intermediate valence of this compound.<sup>374</sup> Investigations of  $\text{YbNi}_2\text{Si}_2$  and  $\text{YbCu}_2\text{Si}_2$  revealed a similar behavior with minima between 50 and 100 K; however, the Seebeck coefficient gets positive for  $\text{YbNi}_2\text{Si}_2$  below  $\sim 20$  K, which was assigned to the antiferromagnetic ordering of this compound.<sup>375,376</sup> High-pressure studies of  $\text{YbCu}_2\text{Si}_2$  revealed that a magnetic phase transition emerges at  $T_m = 1.9$  K around  $T_C = 10.2$  GPa which shifts to higher temperatures with increasing pressure.<sup>377</sup> This was later confirmed by high-pressure measurements of the Seebeck coefficient.<sup>375</sup>  $\text{YbPd}_2\text{Ge}_{1.7}$  also exhibits a nonmonotonous behavior and shows a negative Seebeck coefficient down to low temperatures, where again the sign changes.<sup>378</sup> The solid solutions  $\text{YbCu}_2\text{Si}_{2-x}\text{Ge}_x$ <sup>379</sup> and  $\text{Yb}_{1-x}\text{La}_x\text{Cu}_2\text{Si}_2$ <sup>380</sup> finally were studied with respect to their thermoelectric properties. Materials with large Seebeck coefficients at low temperatures can be utilized for cryogenic Peltier elements for cooling infrared sensors on satellites. With this respect, the solid solutions were investigated to maximize the  $zT$  at cryogenic temperatures. In both cases, an enhancement of the power factor as well as a shift of the maximum to higher temperatures was observed. Alongside this, a reduction of the thermal conductivity was noted, leading to an overall higher  $zT$ . The observed changes were attributed to the changes in the unit cell dimensions that can be envisioned as “chemical pressure”.

$\text{YbAlB}_4$  and  $\text{YbNiAl}_4$ , although having the same chemical composition, adopt two different structure types.  $\text{YbAlB}_4$  is furthermore dimorphic and crystallizes either in the orthorhombic  $\text{YCrB}_4$  ( $\alpha\text{-YbAlB}_4$ ; space group  $Pbam$ ) or  $\text{ThMoB}_4$

( $\beta$ -YbAlB<sub>4</sub>; space group *Cmmm*) type structure (vide ultra). Due to the superconductivity observed in  $\beta$ -YbAlB<sub>4</sub><sup>204</sup> and its heavy Fermion characteristics, the quantum-critical point (QCP) is of special interest. During these investigations, it became clear that the Seebeck coefficient for  $\beta$ -YbAlB<sub>4</sub> behaves significantly different compared to YbRh<sub>2</sub>Si<sub>2</sub>.  $-S(T)/T$  becomes large for  $\beta$ -YbAlB<sub>4</sub> near the zero-field QCP, while for YbRh<sub>2</sub>Si<sub>2</sub>,  $-S(T)/T$  is strongly suppressed.<sup>370,381</sup>

Also, in the CeCr<sub>2</sub>Al<sub>20</sub> type structure (space group *Fd $\bar{3}m$* ), several Yb-containing compounds are known that exhibit Kondo or heavy Fermion properties. Similar to the examples discussed before, the members of the YbT<sub>2</sub>Zn<sub>20</sub> (*T* = Fe, Ru, Os, Co, Rh, Ir) series also exhibit heavy Fermion Kondo lattices. In their thermoelectric properties, all members show large, negative minima (between  $-75$  and  $-45$   $\mu\text{V K}^{-1}$ ); however, variations with respect to the transition metal *T* can be observed. The local minimum in *S*(*T*) and the local maximum in  $\rho$ (*T*) can be correlated to the transition metals and the Kondo temperatures of the respective compounds as well as with the influence of the crystalline electric field splitting.<sup>382</sup> Wei and co-workers have shown, that the compounds of the YbT<sub>2</sub>Zn<sub>20</sub> (*T* = Co, Rh, Ir) series exhibit increasing figures of merit with increasing mass of the transition metals at low temperatures, peaking at  $zT = 0.07$  at 35 K for YbIr<sub>2</sub>Zn<sub>20</sub>.<sup>383</sup>

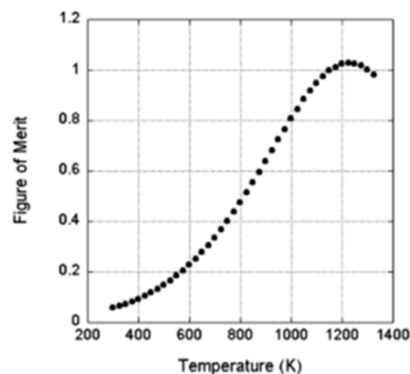
Finally, the thermoelectric investigations on Yb<sub>2</sub>Pt<sub>6</sub>Al<sub>15</sub> (Sc<sub>0.6</sub>Fe<sub>2</sub>Si<sub>4.9</sub> type, space group *P6<sub>3</sub>/mmc*),<sup>384</sup> Yb<sub>2</sub>T<sub>3</sub>Al<sub>9</sub> (*T* = Rh, Ir; Y<sub>2</sub>Co<sub>3</sub>Ga<sub>9</sub> type, space group *Cmcm*),<sup>385</sup> Yb<sub>2</sub>Pt<sub>3</sub>Sn<sub>5</sub> (own type, space group *Pnma*),<sup>386</sup> and Yb<sub>2</sub>Ni<sub>12</sub>P<sub>7</sub> (Zr<sub>2</sub>Fe<sub>12</sub>P<sub>7</sub> type, space group *P6*)<sup>387</sup> should be addressed. While the first two representatives have been characterized as heavy Fermion systems, Yb<sub>2</sub>Pt<sub>3</sub>Sn<sub>5</sub> is a rare example of a heterogeneous mixed-valence system. Yb<sub>2</sub>Ni<sub>12</sub>P<sub>7</sub> finally exhibits a crossover between Fermi-liquid and non-Fermi-liquid behavior and seems to be in close proximity to a quantum critical point. For both Yb<sub>2</sub>Pt<sub>6</sub>Al<sub>15</sub> and the Yb<sub>2</sub>T<sub>3</sub>Al<sub>9</sub> (*T* = Rh, Ir) series, the temperature dependence of the Seebeck coefficient shows a minimum at low temperatures, as expected for Yb-based Kondo lattice systems. While the platinum compound reaches  $-70$   $\mu\text{V K}^{-1}$  around 35 K, Yb<sub>2</sub>Rh<sub>3</sub>Al<sub>9</sub> and Yb<sub>2</sub>Ir<sub>3</sub>Al<sub>9</sub> show a change of sign near 250 K and minima of  $-15$   $\mu\text{V K}^{-1}$  and  $-25$   $\mu\text{V K}^{-1}$  at 30 and 40 K, respectively. For the latter, an additional anomaly is observed near their Néel temperature.<sup>385</sup> A single anomaly in *S*(*T*) is observed for Yb<sub>2</sub>Pt<sub>6</sub>Al<sub>15</sub> indicating that the Kondo energy scale is close to the CEF splitting. The temperature observed is compatible with a Kondo scale of the order of 60 K. The accordance of these temperatures supports the picture of a Kondo lattice with a large characteristic energy in which the whole multiplet is involved in the low-temperature physics.<sup>384</sup> Yb<sub>2</sub>Pt<sub>3</sub>Sn<sub>5</sub> also shows a minimum in the Seebeck coefficient ( $-38$   $\mu\text{V K}^{-1}$  at 60 K),<sup>386</sup> which is typical for intermediate valent Yb compounds such as YbAgCu<sub>4</sub> and YbCu<sub>2</sub>Si<sub>2</sub>.<sup>374</sup>

### Zintl Phases

Zintl phases are usually the class of materials that come to mind when speaking about thermoelectric properties. Due to their semiconducting properties, they can be utilized to produce energy based on a temperature gradient. In many cases, the materials have to be optimized with respect to their electronic properties like charge carrier concentration.

One compound with outstanding thermoelectric properties is probably Yb<sub>14</sub>MnSb<sub>11</sub> (Ca<sub>14</sub>AlSb<sub>11</sub> type, space group *I4<sub>1</sub>/*

*acd*). In 2006, Brown et al. reported that Yb<sub>14</sub>MnSb<sub>11</sub> “achieves quadrupled efficiency and virtually doubled figure of merit over the current state-of-the-art, SiGe”.<sup>388</sup> The material reaches a figure of merit of  $\sim 1.0$  at 1223 K (Figure 17). The Seebeck



**Figure 17.** Figure of merit  $zT$  of Yb<sub>14</sub>MnSb<sub>11</sub>. Maximum  $zT$  value of  $\sim 1.0$  at 1200 K. Reprinted with permission from ref 388. Copyright 2006 American Chemical Society.

coefficient is positive in the whole temperature range and increases linearly with increasing temperature, reaching  $+185$   $\mu\text{V K}^{-1}$  at 1275 K. Since the resistivity also shows a linear behavior with temperature and Hall measurements indicate a constant carrier concentration over the investigated temperature range, the compound can be considered a heavily doped semiconductor. The high figure of merit finally arises from the low thermal conductivity of Yb<sub>14</sub>MnSb<sub>11</sub>, which is on the scale of a glass ( $7\text{--}9$   $\text{mW cm}^{-1} \text{K}^{-1}$ ).

Since the initial report of Yb<sub>14</sub>MnSb<sub>11</sub> being a high- $zT$  material, numerous reports have been published that modified the original compound via the formation of solid solutions as well as investigations on isostructural compounds. In the following part, selected publications have been chosen to illustrate the overall ideas.

Yb<sub>14</sub>MnSb<sub>11</sub> can be modified on the Yb, the Mn, and the Sb position, either by isovalent atoms, often leading to structural distortions and therefore changes in the electronic structure, or by elements which exhibit different formal oxidation states, leading to changes in the overall electron count. Isovalent substitutions on the Yb position can, for example, be realized by Ca. The solid solution Yb<sub>14-x</sub>Ca<sub>x</sub>MnSb<sub>11</sub> shows that the substitution of Yb by Ca reduces the carrier concentration leading to an increased Seebeck coefficient and at the same time to an increased electrical resistivity. In addition, the mass difference between Ca and Yb leads to a significantly reduced density and mass-disorder scattering is observed, resulting in a lowered lattice thermal conductivity. However, these changes do not lead to a significant improvement of  $zT$ . In addition, compositions with values of  $x > 4$  get highly air-sensitive, making the characterization of the thermoelectric properties impossible.<sup>389,390</sup>

As for a substitution of Yb via a trivalent rare earth element, Sc and Y,<sup>391</sup> La,<sup>392</sup> Lu,<sup>393,394</sup> as well as the full RE<sup>3+</sup> (La–Lu) series<sup>395</sup> have been reported. While for the latter only the thermochemistry and thermal stability was analyzed, for the other substitutions the influence on the thermoelectric properties was also investigated since the trivalent rare earth cations change the carrier concentration. The Sc- and Y-substituted specimens show higher resistivities and thus higher

Seebeck coefficients; however, an overall reduction of the  $zT$  is observed.<sup>391</sup> Also for the Lu-substituted compounds, a reduced carrier concentration was reported as well as a reduced peak  $zT$ .<sup>394</sup> Another study, however, reported an increased carrier concentration upon Lu substitution leading to a  $\sim 30\%$  improvement in  $zT$  at 670 K.<sup>393</sup> For La-substituted  $\text{Yb}_{14-x}\text{La}_x\text{MnSb}_{11}$ , finally an increased  $zT$  was reported for  $x \sim 0.4$  with a peak  $zT$  of 1.15 at 1150 K.<sup>392</sup>

Also for the Mn position, an isovalent substitution is possible, e.g., in  $\text{Yb}_{14}\text{Mn}_{1-x}\text{Zn}_x\text{Sb}_{11}$ .<sup>396</sup> The substitution shows a slight improvement up to  $x \leq 0.4$  by reducing the lattice thermal conductivity at low temperatures; however, only a negligible impact at high temperatures is observed. This is based on the reduction of spin disorder scattering leading to an overall improvement of about 10% with a maximum thermoelectric figure of merit  $\sim 1.1$  at 1275 K for  $x = 0.4$ . Substituting  $\text{Mn}^{2+}$  for  $\text{Al}^{3+}$ , however, again leads to a reduction of the carrier concentration,<sup>397,398</sup> similar to the substitution of  $\text{Yb}^{2+}$  by  $\text{RE}^{3+}$ . For  $x = 0.6$  and 0.8, an improvement of 30% was observed, equal to a  $zT = 1.3$  at 1223 K.

Finally, modifications on the Sb positions should be mentioned. In the solid solution  $\text{Yb}_{14}\text{MnSb}_{11-x}\text{Bi}_x$ , overall lower  $zT$  values are observed,<sup>399</sup> while  $\text{Yb}_{14}\text{MnSb}_{11-x}\text{Te}_x$  shows an improved  $zT$  of 1.11 for  $x = 0.07$  at 1240 K.<sup>400</sup> This, again, can be attributed to a reduced carrier concentration.

Furthermore, the low-temperature behavior of the thermoelectric properties has been studied at ambient and increased pressure. Also in the range between 4.2 and 300 K, the Seebeck coefficient for  $\text{Yb}_{14}\text{MnSb}_{11}$  is positive and increases upon heating.<sup>401</sup> Increasing the pressure up to 2.3 GPa leads to a further increase of the Seebeck coefficient in the same temperature range.<sup>402</sup>

Besides  $\text{Yb}_{14}\text{MnSb}_{11}$ , other Yb-containing compounds with the same composition and crystal structure were also investigated. As examples,  $\text{Yb}_{14}\text{MgSb}_{11}$ ,  $\text{Yb}_{14}\text{MnBi}_{11}$ , and  $\text{Yb}_{14}\text{MgBi}_{11}$  will be discussed.  $\text{Yb}_{14}\text{MgSb}_{11}$  has, compared to its Mn-containing analogue, an increased Seebeck coefficient which leads to an increase of 45% in the figure of merit ( $zT = 1.02$  at 1075 K) at the same temperature.<sup>403</sup> The addition of iron leads to the formation of nanosized inclusions, which lead to an even further increase of  $zT$ .<sup>404,405</sup>  $\text{Yb}_{14}\text{MnBi}_{11}$  and  $\text{Yb}_{14}\text{MgBi}_{11}$  are both metallic in nature, and  $\text{Yb}_{14}\text{MgBi}_{11}$  has an increased carrier concentration in comparison to the isostructural antimonide which compensates potential improvements of the thermoelectric properties. Overall, the  $zT$  values of both compounds reach  $\sim 0.2$  at 875 K.<sup>406</sup>

The  $\text{MT}_2\text{X}_2$  compounds that adopt the trigonal  $\text{Ce}_2\text{O}_2\text{S}$  type structure (space group  $\bar{P}3m1$ ) with the general composition  $\text{M}^{\text{II}}\text{T}^{\text{II}}_2\text{Pn}_2$  can also be considered as valence precise Zintl phases.  $M$  is a divalent alkaline earth or rare earth metal (Ca, Sr, Eu, Yb), while  $T$  is a divalent transition metal (Mn, Zn, Cd) or Mg, and  $\text{Pn}$  finally a pnictogen (As, Sb, Bi). These compounds are considered to be “electron precise” according to the Zintl formalism.

In the following paragraph, just a small number of studies will be mentioned as there is a plethora of studies related to the optimization of the thermoelectric properties by different strategies. One of the first reports is on the solid solution  $\text{Ca}_x\text{Yb}_{1-x}\text{Zn}_2\text{Sb}_2$  from Gascoin and co-workers.<sup>407</sup> They showed that  $\text{YbZn}_2\text{Sb}_2$  has the smallest Seebeck coefficient but at the same time the smallest electrical resistivity.  $\text{CaZn}_2\text{Sb}_2$  in contrast exhibits an over two times larger

Seebeck coefficient but also a 7-fold larger resistivity. The solid solution shows an intermediate behavior, depending on the composition. Subsequently, numerous solid solutions were investigated with respect to improving the TEP; for example, substitution of Yb by Eu according to  $\text{Eu}_x\text{Yb}_{1-x}\text{Zn}_2\text{Sb}_2$ <sup>408,409</sup> or by La as in  $\text{Yb}_{1-x}\text{La}_x\text{Zn}_2\text{Sb}_2$ .<sup>410</sup> But also high-quality single crystals of  $\text{YbZn}_2\text{Sb}_2$  were grown and investigated.<sup>411</sup> Despite modifications by chemical substitution, nanoporous<sup>412</sup> or Yb-deficient materials ( $\text{Yb}_{1-\delta}\text{Zn}_2\text{Sb}_2$ )<sup>413</sup> were also synthesized and investigated. But not only the substitution of the Yb is possible. Also, the formation of solid solutions with respect to the Zn or Sb position can influence the thermoelectric properties. For example, the solid solutions  $\text{YbZn}_{2-x}\text{Mg}_x\text{Sb}_2$ ,<sup>414</sup>  $\text{YbZn}_{2-x}\text{Mn}_x\text{Sb}_2$ ,<sup>415,416</sup>  $\text{YbZn}_{2-x}\text{Cd}_x\text{Sb}_2$ ,<sup>417</sup> and  $\text{YbZn}_2\text{Sb}_{2-x}\text{Bi}_x$ <sup>418</sup> should be named. Finally, it has been proven that Mg can substitute on both the Yb and Zn sites.<sup>419</sup> Besides  $\text{YbZn}_2\text{Sb}_2$ , also  $\text{YbMg}_2\text{Sb}_2$ ,<sup>420</sup>  $\text{YbMg}_2\text{Bi}_2$ ,<sup>421–423</sup>  $\text{YbMn}_2\text{Sb}_2$ ,<sup>424</sup>  $\text{YbCd}_2\text{Sb}_2$ , and the solid solutions  $\text{YbCd}_{2-x}\text{Mn}_x\text{Sb}_2$ <sup>425</sup> have been investigated in great detail.

Finally, two other classes of important thermoelectric materials should be addressed: clathrates and skutterudites. Although clathrates are structurally very versatile materials, hardly any ytterbium-containing ones have been reported. Detailed information on the different clathrate structures and their applications can be found in a review.<sup>426</sup> The key features that lead to their intriguing thermoelectric properties are the “rattling” atoms within the structure. These are located in the large polyhedra that form the framework and can effectively scatter the heat-carrying phonons, therefore reducing the thermal conductivity. The reports on Yb-substituted clathrates are mainly examples of type-I clathrates, which crystallize in space group  $\text{Pm}\bar{3}n$  and exhibit the general formula  $\text{G}_8[\text{F}_{46}]$ , with  $F$  being the framework and  $G$  being the guest atoms. In the type-I clathrates, two different polyhedra can be found, a 20-vertex pentagonal dodecahedron and a 24-vertex tetrakaidcahedron. For the framework, different elements and elemental combinations are known as well as a large number of guest atoms occupying these polyhedra.<sup>426</sup> The number of Yb-based type-I clathrates ( $\text{Na}_4\text{Si}_{23}$  type, space group  $\text{Pm}\bar{3}n$ ), however, is extremely low. To the best of our knowledge, no pure Yb clathrates have been reported, but the solid solutions  $\text{Ba}_{8-x}\text{Yb}_x\text{Ga}_{16}\text{Si}_{30}$ ,<sup>427</sup>  $\text{Ba}_{8-x}\text{Yb}_x\text{Ga}_{16}\text{Ge}_{30}$ ,<sup>428</sup>  $\text{Ba}_{8-x}\text{Yb}_x\text{Ni}_{0.1}\text{Zn}_{0.54}\text{Ga}_{13.8}\text{Ge}_{31.56}$ ,<sup>429</sup>  $\text{Ba}_{8-x}\text{Yb}_x\text{Ni}_{3.8}\text{Si}_{42.2}$ ,<sup>430</sup> and  $\text{Ba}_{8-x}\text{Yb}_x\text{Cu}_{4.9}\text{Si}_{41.1}$ <sup>430</sup> have been investigated experimentally. In addition, first principle studies on  $\text{Ba}_6\text{Yb}_2\text{Au}_6\text{Ge}_{40}$ <sup>431</sup> have been conducted. For the silicide, it was shown that the substitution of Ba by Yb is possible, but only to a certain extent of  $x = 0.25$  to 0.38.<sup>427</sup> The substitution led to a slight decrease of the Seebeck coefficient, however, also to a significant decrease of the lattice thermal conductivity which resulted in a slight overall increase of  $zT$  for  $\text{Ba}_{7.75}\text{Yb}_{0.25}\text{Si}_{31.9}\text{Ga}_{14.1}$ .<sup>427</sup> For the germanide series  $\text{Ba}_{8-x}\text{Yb}_x\text{Ga}_{16}\text{Ge}_{30}$ , in principle the same trend is observed. Here, samples with  $x = 0$  to 1.3 could be synthesized and investigated, and again a decrease in the Seebeck coefficient is visible for  $x \leq 0.7$  alongside a decrease of the lattice thermal conductivity leading to a  $zT$  of 1.1 (950 K) for  $\text{Ba}_{7.5}\text{Yb}_{0.5}\text{Ga}_{16}\text{Ge}_{30}$ . This is an increase of 90% compared to the Yb-free sample.<sup>428</sup> Also in  $\text{Ba}_{8-x}\text{Yb}_x\text{Ni}_{0.1}\text{Zn}_{0.54}\text{Ga}_{13.8}\text{Ge}_{31.56}$ ,<sup>429</sup> a slightly increased  $zT$  (15%,  $zT = 0.91$  at 900 K) was reported for  $x = 0.3$ . In this case, the electrical resistivity and lattice thermal conductivity decrease with increasing  $x$ , the Seebeck coefficient, however,

stays constant. Yb substitution in  $\text{Ba}_{8-x}\text{Yb}_x\text{Ni}_{3.8}\text{Si}_{42.2}$  and  $\text{Ba}_{8-x}\text{Yb}_x\text{Cu}_{4.9}\text{Si}_{41.1}$  revealed that the majority of the Yb precipitated as secondary phases. The solubility limit is only about 1% on the Ba site. The multiphase samples showed both a lower electrical resistivity and a lower Seebeck coefficient with respect to the unsubstituted compounds; however, the thermal conductivity increases. Overall, the thermoelectric figure of merit of the Ni-based system is not improved, while a slight enhancement above 800 K is observed for  $\text{Ba}_{8-x}\text{Yb}_x\text{Cu}_{4.9}\text{Si}_{41.1}$ .<sup>430</sup> For  $\text{Ba}_6\text{Yb}_2\text{Au}_6\text{Ge}_{40}$ , finally, the thermoelectric properties were calculated using the full-potential linearized augmented plane wave method based on density functional theory. It was shown that the substitution enhances the density of states at the bottom of the conduction band, leading to an increased Seebeck coefficient due to an increased effective mass of the conduction band.<sup>431</sup>

Finally, one report on a type-VIII clathrate ( $G_8[F_{46}]$ ; own type, space group  $I\bar{4}3m$ ) has been published, which investigated the substitution in  $\text{Ba}_{8-x}\text{Yb}_x\text{Ga}_{16}\text{Sn}_{30}$ .<sup>432</sup> Here, an improvement of the Seebeck coefficient up to  $x = 1.0$  was observed. Overall, the situation is similar to the type-I clathrates: a substitution adjusts the Seebeck coefficient as well as the electrical and thermal conductivity, leading to an improvement of the figure of merit  $zT$ . In  $\text{Ba}_{6.5}\text{Yb}_{1.5}\text{Ga}_{16}\text{Sn}_{30}$ , this leads to a maximum  $zT$  of 1.35 at 489 K.

The last class of materials that should be discussed with respect to their thermoelectric properties are Yb-containing skutterudites. The name of this class of materials derives from the mineral skutterudite  $\text{CoAs}_3$  with the general formula  $\text{TX}_3$  ( $\text{CoAs}_3$  type, space group  $Im\bar{3}$ ). In the crystal structure of the skutterudites, two empty cavities can be found that can accompany a number of different atoms. In the late 1970s, Jeitschko and co-workers coined the term “filled skutterudites” for compounds with the general composition  $\text{MT}_4\text{X}_{12}$  ( $\text{LaFe}_4\text{Sb}_{12}$  type, space group  $Im\bar{3}$ ).<sup>433–436</sup> In 1996, Sales<sup>437</sup> and Fleurial<sup>438</sup> showed that filled skutterudites exhibit excellent thermoelectric properties at elevated temperatures. The reviews by Uher from 2001 and Sales from 2003 summarize the important aspects of skutterudites.<sup>439,440</sup> One of the first Yb-containing filled skutterudites investigated for its physical properties was  $\text{YbFe}_4\text{Sb}_{12}$ . Magnetic measurements indicate an intermediate valence state and a slightly enhanced Sommerfeld coefficient.<sup>441,442</sup> In these reports, the potential for  $\text{YbFe}_4\text{Sb}_{12}$  to exhibit good thermoelectric properties was outlined. Low-temperature thermoelectric property measurements showed that  $\text{YbFe}_4\text{Sb}_{12}$  is a p-doped metal with a low carrier density. The high electrical resistivity leads to a  $zT < 0.02$  below 300 K.<sup>443</sup> The investigations of the high-temperature properties lead to a  $zT = 0.4$  at 670 K.<sup>444</sup> Within the solid solution  $\text{Yb}_y\text{Fe}_{4-x}\text{Ni}_x\text{Sb}_{12}$ , even  $zT$  values near 1 (750–850 K) could be obtained for  $x = 0.6–0.8$ .<sup>445</sup>

Due to the chemical versatility of the filled skutterudites, numerous compositions were investigated with respect to their thermoelectric properties. Within the field of Yb-based compounds,  $\text{YbCo}_4\text{Sb}_{12}$  showed an excellent performance, leading to a large number of studies of tuned and improved materials. Therefore, only some examples will be summarized in the next paragraph. Studies on different compounds of the  $\text{Yb}_x\text{T}_4\text{Sb}_{12}$  series ( $x = 0.1–0.8$ ,  $T = \text{Fe, Co, Rh, Ir}$ ) showed that the physical properties seem to correlate with the filling  $x$  and the electronic situation modified by changing  $T$ .<sup>446</sup> At the same time, studies on  $\text{Yb}_y\text{Co}_4\text{Sn}_x\text{Sb}_{12-x}$  indicated a maximum  $zT$  of 0.1 at 300 K for  $\text{Yb}_{0.44}\text{Co}_4\text{Sb}_{12}$ .<sup>447</sup> More detailed studies

including Hall effect, electrical resistivity, Seebeck coefficient, and thermal conductivity measurements were conducted on the solid solution  $\text{Yb}_y\text{Co}_4\text{Sb}_{12-x}\text{Sn}_x$  for  $y = 0.19$  and 0.5. It was shown that substituting Sn on the Sb site does not alter the Fermi level with respect to the conduction band; instead, a p-type band for a sufficiently large Sn content  $x$  was observed. Therefore, this is not a valid method to optimize the thermoelectric properties of Yb-filled skutterudites.<sup>448</sup> Double substitution according to  $\text{Ba}_{0.08}\text{Yb}_{0.09}\text{Co}_4\text{Sb}_{12}$ , however, led to a broad range of resonant phonon scattering and which resulted in a strong suppression in the lattice thermal conductivity. Therefore, high  $zT$  values of 1.36 at 800 K could be observed.<sup>449</sup> A similar value in  $zT$  (1.36 at 850 K) could be observed for  $\text{Al}_{0.3}\text{Yb}_{0.25}\text{Co}_4\text{Sb}_{12}$ .<sup>450</sup> By the help of ball-milling in combination with hot-pressing, the value of  $x$  in  $\text{Yb}_x\text{Co}_4\text{Sb}_{12}$  could be improved to almost 0.5, leading to a  $zT$  of 1.2 at 823 K.<sup>451</sup> In recent years, studies addressing the band conversion in  $\text{Yb}_x\text{Co}_4\text{Sb}_{12}$ ,<sup>452</sup> the achievement of ultralow thermal conductivities in  $\text{Yb}_x\text{Co}_{4-y}\text{Fe}_y\text{Sb}_{12}$  with  $x = 0.25–0.5$  and  $y = 0.1–0.5$ ,<sup>453</sup> the synthetic approach on  $\text{Yb}_x\text{Co}_4\text{Sb}_{12}$ ,<sup>454</sup> and introducing nano-inclusions<sup>455</sup> have led to a deeper understanding of the possibilities to tailor the thermoelectric properties in skutterudite materials.

## SUMMARY AND OUTLOOK

The present review impressively underlines that the family of ytterbium-based intermetallic compounds covers extremely broad facets of crystal structures along with interesting and, in some cases, even outstanding physical properties. This interplay is mostly driven by the ytterbium valence: divalent ( $4f^{14}$ ), intermediate-valent, or trivalent ( $4f^{13}$ ) ytterbium. Thus, one observes simple diamagnets, Pauli or Curie–Weiss paramagnets as well as valence fluctuating compounds, Kondo behavior, heavy Fermion ground states, or sometimes even superconductivity.

Some of the trivalent ytterbium intermetallics have gained deep interest due to these physical properties. The phases have been meticulously studied with respect to their properties using magnetic and transport measurements, specific heat data, NMR, ESR, and Mössbauer spectroscopy, elastic and inelastic neutron scattering, XAS and XPS data, as well as detailed thermoelectric measurements, then allowing the development of structure–property relations.

Among the binary Yb intermetallics,  $\text{YbAl}_2$  and  $\text{YbAl}_3$  have received the most attention due to their Kondo behavior. Overall, however, the number of binary compounds studied in detail is rather low compared to the ternary systems, given the significantly larger combinatory manifold in the ternary systems. Here,  $\text{YbRh}_2\text{Si}_2$  is the most outstanding phase with >350 publications on this compound. It exhibits a non-Fermi-liquid behavior along with antiferromagnetic ordering at  $T_N = 70$  mK. A quantum critical point at ambient pressure is suggested, at which the magnetic ordering can be suppressed by a magnetic field of only  $B_c = 60$  mT.  $\text{Yb}_2\text{Pt}_2\text{Pb}$  exhibits a Shastry–Sutherland lattice that causes a high anisotropy in the magnetic susceptibility with an about 30 times larger value for  $\chi_{(100)}$  as compared to  $\chi_{(001)}$ . Finally, among the series of  $\text{YbT}_2\text{Zn}_{20}$  compounds, the cobalt-containing representative  $\text{YbCo}_7\text{Zn}_{20}$  exhibits Kondo interactions, and from specific heat measurements, a large Sommerfeld coefficient of  $\gamma = 7.9 \text{ J}^{-1} \text{ mol}^{-1} \text{ K}^{-2}$  was deduced. Finally,  $\text{Yb}_{14}\text{MnSb}_{11}$  is one of the most promising p-type thermoelectric materials with a high  $zT > 1$ , making this compound suitable for different applications.

Although most of these studies are located in basic research, they help with a deeper understanding of complex intermetallic compounds and motivate for future studies. Especially, the ytterbium-based thermoelectric materials might find application. Even though the family of ytterbium intermetallics is large, a weak point concerns the overall systemization of the property data, since only few larger isotypic series of compounds exist. Already, the equiatomic YbTX phases crystallize with more than 10 different structure types; many of them are superstructures of the aristotype AlB<sub>2</sub>. The probably best investigated, and thus comparable, series of isotypic compounds concerns the CeCr<sub>2</sub>Al<sub>20</sub> type zinc compounds YbT<sub>2</sub>Zn<sub>20</sub> with T = Fe, Co, Ru, Rh, Os, and Ir.

The study of ytterbium intermetallics is still a vivid field in solid state chemistry and physics. Besides the discovery and characterization of new phases, the understanding and systemization of the structure–property relations is an important future task.


## ■ ASSOCIATED CONTENT

### Data Availability Statement

The data underlying this study are available in the published article.


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